5101/232 Flat Plate Solar Array Project

(WASA-CR-173460) CHERICAL ECADING TECHNOLOGY FOR TERRESTRIAL PROTOVOLTAIC MODULES (Jet Propulsion Lab.) 62 P HC AD4/MF A01 CSCL 10A

N84-22008

Unclas G3/44 18991

Chemical Bonding Technology for Terrestrial Photovoltaic Modules

Status to February, 1983

O. R. Coulter

E. F. Guddiny

E. P. Placedoman



November 15: 1983

Prepared for

U.S. Department of Energy

Through an Agreement with National Aeronautics and Space Administration

Os

Jet Propulsion Laboratory Galifornia Institute of Technology Pasadena, Çalifornia

JPL Publication 83-86

Chemical Bonding Technology for Terrestrial Photovoltaic Modules

Status to February, 1983

D. R. Coulter E. F. Cuddiny E. P. Plueddeman

November 15, 1983

Prepared for

U.S. Department of Energy
Through an Agreement with
National Aeronautics and Space Administration
by
Jef Propulsion Caboratory
California Institute of Technology
Pasadena, California

JPL Publication 83-86.

Prepared by the Jet Fropulsion Laboratory, California Institute of Technology, for the U.S. Department of Energy through an agreement with the National Aeronautics and Space Administration.

The JPI. Flat-Plate Solar Array Project is sponsored by the U.S. Department of Energy and is part of the Photovoltaic Energy Systems Program to initiate a major effort toward the development of cost-competitive solar arrays.

This report was prepared as an account of work sponsored in part by an agency of the United States Government. Neither the United States Government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights.

Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise, does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.

This publication reports on work done under NASA Task RE-152, Amendment 66, DOE / NASA IAA No. DE-AI01-76ET20356.

ABSTRACT

Encapsulated photovoltaic modules must hold together for 20 years, reliably resisting delamination and separation of any of the component materials. Delamination of encapsulation materials from each other, or from solar cells and interconnects, can create voids for accumulation of water, promoting corrosive failure. Delamination of silicone elastomers from unprimed surfaces was a common occurrence with early modules, but the incidences of silicone delamination with later modules decreased when adhesion promotera recommended by silicone manufacturers were used. An investigation of silicone delamination from unprimed surfaces successfully identified the mechanism, which was related to atmospheric oxygen and moisture. This early finding indicated that reliance on physical bonding of encapsulation interfaces for long life in an outdoor environment would be risky.

For long outdoor life, the material components of a module must therefore be held together by weather-stable adhesion promoters that desirably form strong, interfacial chemical bonds. The Environmental Isolation Task of the Flat-Plate Solar Array Project (FSA), managed by the Jet Propulsion Laboratory (JPL) for the U.S. Department of Energy (DOE), conducts a program to identify, develop, and validate weather-stable chemical bonding adhesion promoters for terrestrial photovoltaic modules. This program is a joint effort currently involving Dow Corning Corp., Case Western Reserve University, Rockwell International, Springborn Laboratories, Inc., and JPL.

This report, the Task's second on chemical bonding technology, is intended to accomplish three purposes:

- (1) Provide a status report on chemically bonding adhesion promoters, with data on bond strength performance.
- (2) Describe and discuss the theories of the chemically bonded interface.
- (3) Identify processing considerations relevant to achieving long-term interfacial chemical bonding.

rous out

This report on chemical bonding technology for terrestrial photovoltaic modules includes information on the current status of chemical bonding technology, a description of the theories of the chemically bonded interface, and an introduction to processing considerations relevant to achieving long-term interfacial chemical bonding. It has been prepared at the Jet Propulsion Laboratory, Californic Institute of Technology, by Daniel R. Coulter and Edward F. Cuddiny, members of the Materials Science Group of the Applied Mechanics Technology Section, and Edwin P. Plueddemann of the Dow Corning Corporation.

PRECEDING PAGE BLANK NOT FILMED

CONTENTS

**		
	۸.	BACKGROUND
	8.	BASIC DEFINITIONS
11.	ADMES	IVES, PRIMERS, AND ADHESION-PROMOTING TECHNIQUES
	A.,	INTRODUCTION
	N.,	EVA PRIMERS AND ADHESIVES
	C.	EMA PRIMERS AND ADRESIVES
	D.	PhBA PRIMERS AND ADMESIVES
	L.	POLYURETHANE (Z-2591) PRIMERS AND ADRESIVES 3
	***	ADDITIONAL MOTES
XXX.	THEOR	IES OF THE CHEMICALLY BONDED INTERFACE
	*A.,	INTRODUCTION TO INTERFACIAL BONDING THEORY
	8.,	PRINCIPALS OF PHYSICAL ADMESION
	C.	CHEMICALLY BONDED INTERFACES
	D.	INTERPHASE STRUCTURE
IV.	PROCE	SSING CONSIDERATIONS RELEVANT TO INTERFACIAL BONDING 5
	Ä×	SURFACE PREPARATION
	В.	APPLICATION
	C.	CURE 1
KEFER	ENCES	
<u>Figur</u>	2 £	
	**	Flat-Plate Solar Module Design Classification
	2.	Material Construction Elements and Interfaces in Photovoltaic Encapsulation Systems

		in Contact With Each Other: (a) in the Absence of Adhesive Material, Leaving Gaps That Can Fill With Air or Contaminants, and (b) with Gap-Filling	
		Adhesive Fresent	%
	4.	Schomatic Illustration of Chemically Bonded Interface	6
	.	Diagram of a Liquid Droplet on a Solid Surface, Showing the Contact Angle (the Angle Between the Solid Surface and a Line Drawn Tangential to the Droplet's Surface at the Joint Where It Touches the Solid Surface)	33
	6.	Polycibylene	36
	7	Prediction of Interface Stability for Fe $_2$ 0 $_3$ Substrate in a Wet Environment as a Function of γ_p and γ_p of Potential Adhesives; Calculation Was Made Using Equation 3 and the Values of 107 mj/m 2 and 1250 mJ/m 2 for γ_p and γ_p of Fe $_2$ 0 $_3$	40
	8.	Formation of Polysiloxane on Glass Surface	42
	9.	Variation in Total Performance of Organic-Inorganic Composites With Change in Volymer Morphology Near the Interface	44
	10.	Sandwich Model of Silane-Primed Glass-EVA Joint	45
	11.	Glass Transition Temperatures of Silane Coupling Agents .	47
	12.	Silane Coupling Agent on Glass: (a) Schematic Representation Showing Chemically Bonded Interface, Strongly Chemisorbed Region and Weakly Physisorbed Region; (b) Plot of the Mechanical Strength of Various Regions Near the Interface, Showing Weakness in the Physisorbed Bulk Silane	48
	13.	Schematic Representation of the Interdiffusion Model for a Silane-Primed Glass-EVA Joint; Open Circles Indicate Regions of Coupling Agent, Filled Circles Indicate	49
	3. 61	Regions of EVA	** 7
	i da u	Plot of Silane-Adhesive Concentration Gradient as a Function of Distance From Inorganic Interface	50
Tables	3		
	1.	Status of Primers, Adhesives, and Adhesion-Promoting	ë s
		The salar and salar and the salar and salar and salar and the salar and	3 3

2∗	Primers, Adhesives, and Adhesion-Promoting Techniques	· '%	*	12
3.4	Adhesive Bond Strengths for EVA (A-9918) Ponded to Various Materials	· «	*	. 19
	Adhesion of EVA (A-9918) to Solder Using Primer No. 16, Showing Importance of Surface Cleanliness for Solder Bonding	. *	*	22
٥,	Adhesion of New, Experimental A-15295 EVA to Sunadex Glass and Mild Steel	× .9	*	22
6.	Adhesive Bond Strengths for EMA Bonded to Various Materials	i e	.*	23
% x	Adhesive Bond Strengths for PnBA (BA-13870) Bonded to Various Materials	ş: #.	**	24
.8 ,	Adhesive Bond Strengths of Z-2591 Polyurethane to Various Materials	i și	*	25
9.	Surface Energy Dispersion and Polar Force Components of Polymerized Silane Coupling Agents With Structure R-Si(OCH ₃) ₃	§ ¥	' ₩-	in in
10.	Isoelectric Points of Metal (Oxide) Surfaces in Water	F 3x	: · ·	38
	Typical Commercial Coupling Agents and Recommended Applications	x. 4t	:#	43

SECTION 1

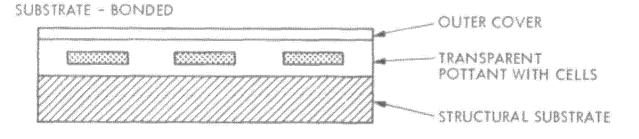
THEROPUSELON

A. BACKGROUND

Photovoltais (PV) modules contain strings of electrically interconnected solar cells Lapable of producing practical quantities of electricity when illuminated by sunlight. Silicon solar cells are fragile and are especially sensitive to brittle failure in tension and bending. The electrically conductive metallization materials (functioning as grids, interconnects, bus bars, and terminals) are sensitive also to excessive flexing, corrosion and other interactions with the terrestrial environment. To ensure the durability of PV modules, their silicon solar cells must be mechanically supported, and the electrically conductive metallization materials must be isolated from environmental exposure.

Encapsulation materials are defined as all construction materials (excluding cells and electrical conductors) required in a PV module to provide mechanical support and environmental isolation. Early efforts to identify a single material that could satisfy all of the encapsulation requirements and needs were unsuccessful (References 1 and 2). The understanding evolved that more than one material would have to be assembled in a composite package to fabricate an encapsulated module satisfying the requirements.

After an examination of all commercial and experimental flat-plate module designs, it was observed that they could be separated into two basic classes (Figure 1). These are designated as substrate-bonded and superstrate-bonded designs, referring to the method by which the solar cells are supported



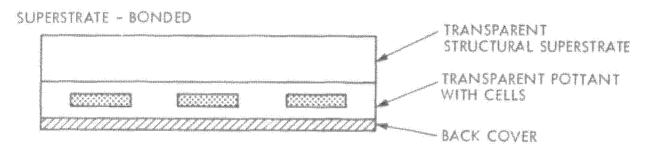


Figure 1. Plat-Plate Solar Module Design Classification

ORIGINAL PAGE IS OF POOR QUALITY

mechanically. In the substrate design, the cells are supported mechanically by a structural substrate, and in the superstrate design, the cells are supported by a transparent structural superstrate. From these two design options, eight basic encapsulation construction elements can be identified. These are illustrated in Figure 2, with their designations and encapsulation functions. Fabricated modules do not need to use all eight of these construction elements, but combinations of these basic elements are incorporated in most module designs. Materials identified for each of these construction elements are listed below:

(1) Low-Soiling Surface Materials:

Fluorinated eilane, L-1668 (3M Co.)

Perfluorodecanoic acid with Dow Corning Corp. Z-6020 primer, E-3820

(2) UV-Screening Front Covers:

Low-iron, tempered float glass (e.g., Sunadex glass, ASG)

UV-screening PMMA accylic films (3M):

Acrylar X-22416, 2 mile thick

Acrylar X-22417, 3 mils thick

UV-screening PVF rluorocarbon film (Du Pont Co.)

Tediar 100BG30UT, 1 mil thick

MODULESUNSDE	LAYER DESIGNATION	ENCAPSULATION FUNCTION
	SURFACE (1) MATERIAL (2) MODIFICATION	EASY CLEANABILITY, ABRASION RESISTANT, ANTIREFLECTIVE
	FRONT COVER	UV SCREENING, STRUCTURAL SUPERSTRATE
	POTTANT	SOLAR-CELL ENCAPSULATION
	POROUS SPACER	AIR RELEASE, MECHANICAL SEPARATION
	DIELECTRIC	ELECTRICAL ISOLATION
S	SUBSTRATE	STRUCTURAL SUPPORT
Section of the sectio	BACK COVER	MECHANICAL PROTECTION, WEATHERING BARRIER, INFRARED EMITTER

PLUS NECESSARY PRIMER-ADHESIVES

Figure 2. Material Construction Elements and Interfaces in Photovoltaic Encapsulation Systems

(3) Potentine

Ethylene vinyl acetate (EVA) (A-9918, Springborn, Du Pont, and Rowland, Inc., Berlin, Connecticut)

Ethylene methyl acrylate (EMA) (A-11877, Springborn)

Poly-n-butyl acrylate (PmBA) (BA-13870, Springborn)

Polyurethane (PU) (2-2591, Development Associates, Inc., North Kingston, R.I.)

(4) Porous Spacer:

Craneglas non-woven E-glass mat, type 230, 5 mils thick (Electrolock, Inc., Chagrin Falls, Ohio)

(5) Dielectric Films:

Candidates are the front-cover and back-cover plastic films

(6) Substrates:

Mild steel

Hardboards

Super-Dorlux, 1/8-in thick (Masonite)

Duron, 1/8-in thick (U.S. Gypnum)

(7) Back Covers (White-Pigmented Plastic Films):

Tedlar 1508L 30 WH, 1.5 mils thick (Du Pont)

Scotchpar 200P WA. 2.0 mils thick (3M Co.)

Korad 63000 White, 3.0 mils thick (Xcel Corp.)

(8) Edge Seal and Gasket:

Butyl edge sealing tape (5354, 3M)

EPDM gasket material (E-633, Pawling Rubber Co., Pawling, N.Y.)

Figure 2 also illustrates the interfaces between adjacent construction elements that will be generated in assembled modules. All of these interfaces in encapsulated modules must hold together for 20 years, reliably resisting partial delamination or total separation. If, for example, delamination of the pottant from solar cells or interconnects occurs, then voids will be formed where liquid water can accumulate, leading to corrosive failure. As another illustration, if back covers are adhesively attached to wood substrate panels to provide humidity protection, and if an adhesion failure were to occur at this joint resulting in a separation of the back cover, then humidity protection for the wood would be lost.

Modules manufactured in the early 1970s generally used silicone elastomers as the pottant material, and delamination from unprimed substrate surfaces was a common occurrence. However, the incidences of silicone delamination virtually ended when adhesion promoters recommended by the silicone manufacturers were used. In the absence of primers or other adhesion promoters, adhesion of the silicone elastomers to the substrate surfaces

involves physical bonding only. A study (References 3 and 4) of silicone delamination from unprimed substrate surfaces identified the mechanism, which was related to atmospheric oxygen and moisture. These results indicated that reliance on purely physical bonding of encapsulated interfaces for long life in an outdoor environment would be risky.

An alternative to physical bonding for interfacial adhesion is chemical bonding (References 5 and 6), where primary chemical bonds bridge the interface between two adjacent materials and hold them together. The adhesion promoters referred to above were formulated to generate such interfacial chemical bonds, resulting in dramatic improvement in resistance against weather-induced delamination.

It therefore became a major FSA activity to identify, develop, and validate durable and weather-stable interfacial chemical bonding techniques for encapsulated photovoltaic modules. The first FSA report on chemical bonding technology (Reference 7) was published in September, 1979, and this, the second report, is a status update. It is divided into four sections: Section I briefly presents definitions and essential descriptions of chemical bonding for readers not familiar with the subject; Section II is an updated listing of chemical bonding agents and an account of performance testing to date; Section III describes modern theories of physical and chemical adhesion, and Section IV presents briefly some of the processing considerations relevant to the formation of strong durable interfacial bonds in PV modules.

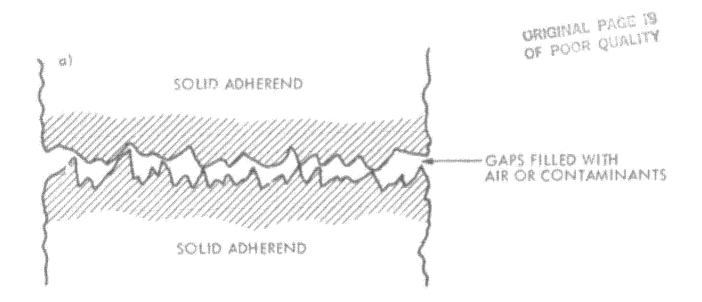
B. BASIC DEFINITIONS

Chemical bonding materials can be divided into three classifications: adhesives, coupling agents, and primers, defined as follows:

Adhesive: A gap-filling material between two rigid (solid) adherends (Figure 3) such as steel and wood, glass and plastic films, etc. The adhesive, when applied, is fluid or viscous to facilitate filling of the gap under some pressure, and to achieve uniform spread over the surfaces of the solid adherends. The adhesive will later lose its fluid flow properties and develop mechanical toughness, either by cooling, if a hot melt, or by being cured to become a thermosetting elastomer or resin. Interfacial adhesion at the adhesive-adherend interface can be established by attracting forces that are physical in nature (physical bonding), by molecular penetration, and/or by primary chemical bonds bridging the adhesive-adherend interfaces.

Coupling Agent: A material that is positioned at the adhesive-adherend interface, and that is chemically reactive with both the adhesive and adherend to generate primary chemical bonds bridging the adhesive-adherend interface (see Figure 4). For two different solid adherends with a common gap-filling adhesive between them, two different coupling agents matched to each adhesive-adherend interface may be needed. In theory, a coupling agent need only be a monomolecular-thick layer between the adhesive and adherend, as its function is chemical bridging.

Primer: A coupling agent dissolved in a volatile solvent, usually at a very low concentration (e.g., 1 to 5 wt %), to achieve a very thinly



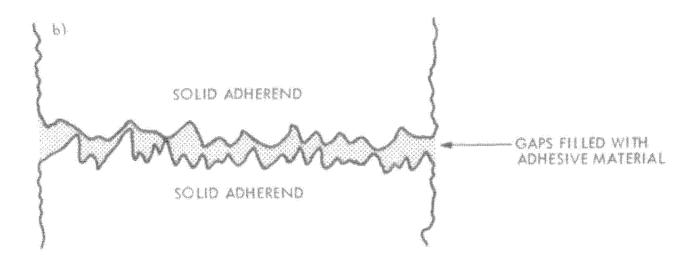


Figure 3. Representation of Two Non-Conforming Solid Surfaces in Contact With Each Other: (a) in the Absence of Adhesive Material, Leaving Gaps That Can Fill With Air or Contaminants, and (b) with Gap-Filling Adhesive Present

deposited layer of the coupling agent on a surface. Sometimes coupling agents need catalysts, reaction promoters, or other chemical compounds to facilitate interfacial chemical bonding. These necessary ingredients are also dissolved in the primer solvent at a concentration chemically related to the coupling agent. (As shown in Section III, coupling agents based on silane chemistry are being investigated for FSA encapsulation interfaces. In many cases water is needed as a reaction promoter and consequently will be used as a primer ingredient in many systems.)

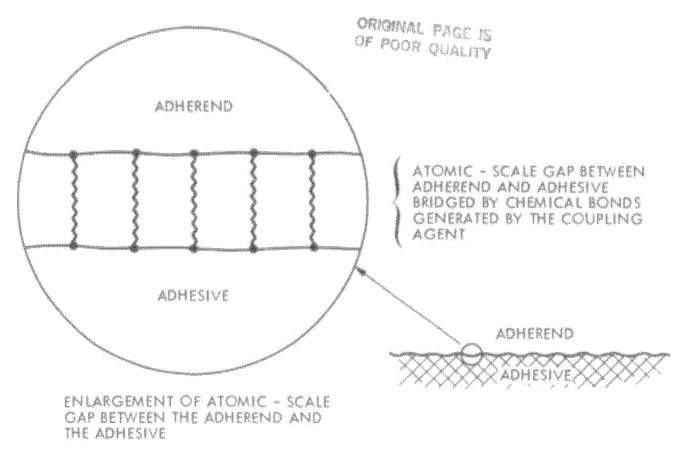


Figure 4. Schematic Illustration of Chemically Bonded Interface

In practice, it could be a property of an adhesive and an adherend to form chemical bonds across their interface naturally, but if it is not, then a properly selected primer solution must be used. If such a solution is used, it is a usual practice to deposit it on the solid adherend surface(s). Alternatively, the coupling agent (and its necessary ingredients) could be dispersed into the adhesive material, to generate a self-priming adhesive. When such an adhesive is applied to an adherend, the coupling agent migrates to the interface to establish the interfacial chemical bonding.

Based on the foregoing definition of an adhesive, a pottant in a PV module can be considered to be such a material, in the sense that the pottant fills the gap between solid front and back encapsulation materials and that it develops mechanical toughness during the heating stages of module fabrication. If a pottant and an adherend such as glass do not naturally form interfacial chemical bonds, then an appropriate primer needs to be identified or developed. As shown in Figures 1 and 2, the pottant in a module achieves mutual contact with the majority of all other encapsulation materials as well as the solar cells and associated electrical circuitry. FSA emphasis in identification and development of primers is directed therefore to the pottant. Since the pottant material, in its current use, with all of its associated primers, has been found to be an effective adhesive, it is logical to examine the potential of this material for adhesive application in other areas of the module. For example, an adhesive is needed to bond white-pigmented plastic-film back covers

to a mild-steel panel. If primers have been identified or developed for chemically bonding a pottant to plastic-film front lovers, and to solar-cell metallization, then the pottant and these primers can be investigated as an adhesive and primer system for the back covers and mild steel. As noted in Section III, this approach has worked for several back-cover and panel interfaces.

Sometimes the surface of a solid adherend does not have the proper chemistry to form chemical bonds with a chemically reactive adhesive, primer, or other coating. This occurs in PV applications with the low-soiling surface coatings L-1668 and E-3620 (p. 2). These two coatings contain chemical groups that will react readily with the surface of glass to form chemical bonds for strong, durable attachment. These same coatings, however, behave differently with Acrylar and Tedlar plastic films. It was found that the chemical attachment of E-3820 to Acrylar, and that of L-1668 to Tedlar, was weak (Reference 8). Therefore, for these specific coatings, the surfaces of the plastic films were treated with ozone to generate polar groups for enhanced chemical reactivity and therefore enhanced chemical attachment. This technique worked well. This adhesion-promoting technique, called surface activation, has only been needed so far for low-soiling coatings on plastic films.

Mechanically, the strength of an adhesive bond is measured under dry conditions, but in outdoor applications, the real assessment of an adhesive bond results from measurements of its bond streng under wet conditions. When wet, the simple criteria of bond quality are that the bonded parts do not readily separate, and that there be a measurable bond strength, greater than zero. Generally, wet bond are igth will be lower than dry bond strength, which is not a concern as long as the wet bond strength is sufficient to hold the parts together against the stress encountered in service. Of course, the best condition would be that both wet and dry failure are cohesive.

For evaluating the durability of a chemically bonded interface, replicas of the bonded system can be immersed in water at room temperature, and the peel strength of a wet sample measured periodically. A clear failure is ready separation of the bonded parts (zero peel strength); success is considered to be conesive failure, even after seven days. The gray area for assessing durability and service-life expectation relates to adhesive failure in which the level of peel strength is intermediate between zero and the cohesive value. There is strength to the bond, but it is difficult to use the data in predicting outdoor life because among other things, it is not known how the bond strength values will change over a 20 year life.

Experience with bond durability indicates that under wet conditions, or exposure to moist atmospheres at high temperatures and humidities, the strength of the bonded interface generally decays logarithmically, with the rate influenced by stress, temperature and relative humidity. The strength of the bonded interface, however, recovers reversibly as environmental conditions become drier, and begins to decay again as moist conditions return. The bond strength does not appear to undergo cumulative damage with such cycling as would be encountered outdoors where weather patterns cycle between wet, moist, and dry conditions.

SECTION 11

ADRESIVES, PRINCES, AND ADRESION-PROMOTING TECHNIQUES

INTRODUCTION Air

This section tabulates all of the adhesives, primers, and adhesionpromoting techniques being evaluated for FSA encapsulation interfaces. In this chapter adhesives, primers, and adhesion promoting techniques are referred to collectively as bonding agents.

The sequence that has been adopted for initially qualifying a bonding agent consists of measuring the dry bond strength (peel strength) of control specimens, then measuring the bond attength of identical specimens that have been immersed in room-temperature water and/or boiling water for various periods. A bonding agent is rejected if its control or its wet value is zero. For some bonding agents, it will be noted that the dry control values are reported, but not the wet strengths; these values were yet to be determined at the time of this publication, and hence an accept-or-reject decision has yet to be made. They are to be considered as promising candidates.

A bonding agent that passes this initial qualification test is considered an experimental candidate that will then be subjected, in bonded test specimens, to accelerated and/or abbraviated aging tests involving ultraviolet light (UV), atmospheric oxygen, humidity, elevated temperatures, and cyclical mechanical and thermal stressing. Aging tests have not yet begun; hence, the bonding agents listed herein are to be considered experimental.

It should also be remarked that the magnitudes of the dry (control) and wet bond-strength values have not been a factor in the initial qualification rating. One reason is that in some cases more than one bonding agent has been identified for a considerable variety of interfaces, which yields a range of bond strengths and therefore creates a temptation to favor the bonding agent with the highest overall dry and wet values. But as none of the bonding agents has yet been aged over long times, under field exposure conditions, there is concern that the initial best performer may age the worst. A second reason is that some of the interfaces are very difficult to bond chemically; therefore, achievement of any measurable dry and wet bond strengths is an accomplishment. For these, it is hoped that the bond strengths will be found adequate for the engineering loads, as well as resistant to change from weathering. The third reason is that, in some cases, the cohesive strength of an adhesive or adherend is low, and therefore this value becomes the maximum bond strength value that can be measured in a mechanical peel test. An example is the poly-n-butyl acrylate (PnBA) pottant, which exhibits cohesive failures within the range of 3 to 8 lb/in. of width (peel test). Such low cohesive strength prevents mechanical measurement of the actual interfacial bond strength, which may be higher, but cannot be measured mechanically because the strength limitation resides in the PnBA. A novel chemical approach (Reference 9) to investigating interfacial bond strengths without this mechanical limitation is discussed in Section III.



Table I lists vertically and horizontally all of the encapsulation materials listed on pp. 2 and 3 and includes metals associated with solar cells and interconnects. All combinations of materials and metals are generated by this matrix format; therefore, combinations that are not seen as practical or possible are identified by an X in the common box generated by the crossing of the row and column of the respective materials or metals. Furthermore, the boxes containing X's are shaded for contrast. The common locations of all other interfacial combinations seen as practical or possible are unshaded, and identified with a letter, or left blank.

The letter S indicates that the interfacial combination requires no primers or adhesives; that is, they bond chemically during module fabrication. This has occurred with only certain combinations of the low-soiling coatings and other cover plastic films, as indicated in Table 1.

The letter B indicates that a bonding agent, primer, adhesive, or adhesion-promoting technique has been either developed or commercially recommended. There are 22 such bonding agents under evaluation; these are listed and described in Table 2, which also includes, where known, their chemical formulations. In this report the bonding agents will be referred to by their assigned numbers in Table 2, 1 to 22, prefixed with the relevant bonding-agent description (primer or adhesive, whichever applies). For example, primer No. 2 is the primer for bonding EVA to glass, and adhesive No. 18 is an adhesive system for bonding Scotchpar plastic film to mild steel. A word of caution about Table 2: The initial development or commercial recommendation of a primer or adhesive arises from the identified need for a specific interfacial combination (e.g., EVA and glass). Once available, however, such a system can be evaluated for use with many other interfacial combinations, often with success. The description of the primer or adhesive system given in Table 2 is a statement of the initial need for the system, but should not be construed as being its only use. Lastly, primers No. 4, 10, 11, and 19 (Table 2) were very recent commercial recommendations, and therefore have not yet been evaluated by FSA.

A blank in Table 1 indicates that the natural bonding behavior and/or needs and requirements for bonding agents for those interfacial combinations has not yet been experimentally determined.

of all of the interfacial combinations noted in Table 1, only those established by the low-soiling coatings on the outer covers cannot be monitored by mechanical peel testing. A different procedure is therefore followed for these coatings. Water poured onto surfaces coated with these materials will quickly bead up into small, discrete droplets, and on tilted surfaces, will readily drain off, in contrast with uncoated outer cover surfaces on which water will spread into a film and will resist run-off when the surfaces are tilted. In addition, the low-soiling property reduces natural soil accomulation as compared with uncoated outer cover surfaces.

Outer-cover materials, coated with all three low-soiling coatings, and uncoated controls were mounted outdoors for natural exposure to weather and soil (References IO and II). The FC-721 and FC-723 coatings were, within a few weeks, totally lost from the surfaces as evidenced by both water-spread

Table 1. Status of Primers, Adhesives, and Adhesion-Promoting Techniques for Encapsulation Interfaces

				£-	466																	
A.	Low-Soiling Coatings	1 .	L+1868		T.	-9.80	9															
		2.4	£+3820	X		180	- 7 d	177														
		3.	TC-721/723	X			55	0.30	6×		isis											
8.	Outer Covers	1,	Sunadex glass	S	8			Ä.	271	27	2-2	261	6/7									
	(Transparent Overlays)	2.	Acrylan X-22636/7	3	75				7.	di.	ik I	00.5	(1)									
		3.	Tedlar 1008G30UT		S			ľx			(hy)	ede	V	ny l	Ac	et.						
C.	Pottants	1,*	Ethylene vinyl acetate	X	X	X						hy I	ene	Me	thy	1	Acrelate					With and
	(Transparent Elastomers)	2.	Ethylene methyl actylate	×				8		X			Iv-	p in E	ig to	1	Korylate					4 4
		3.	Poly-n-butyl acrylate	X	X	X			35	X	X		Po	Lvu	ret	t a	ne Z-2591					
		4,	Polyurethane Z-2591			X		8		X	X			183	18	St						
D.	Substrate Panels	1.	Mild steel	×		X	×		X							rđ	oo a ref					A1100
		2.4	Hardboard				X	B									edlar 150		NAME.			
E.	Back Covers/Dielectric	1.	Ted1ar 150BL30WH	X	X	*	×	X	X								Sentich	er.	ocr	224		E A
	Films (White-Pigmented	2.	Scotchyar 2007 VI	X	×	×	X	X				B				Х	Kori	id 63	1000	(86)	tel	
	Films)	3.	Korad 53000 (White)	×	Х	X		7						13		X		old:		m v.		
	Solar Cell Metallization	1.	Solder	×	×	X	×	X		.8				×	X	X		0.0	1000	*		
		7.	Copper	X	×								8	K	X					cke1		
		3.	*ickel	×	X.	7	X		X				P	×		×				Alop	ine	
		4.	Alminus	X	X	X		1					10	X		X				S	(lver	
		5.	Silver	×	X		X	X						X				1		x i	Czaner	Las
6.	ElectMech. Spacer	1.	Crasexlas	X			×	×	X					Х		×						

Symbols:

- 1. S. * Self-bonding interface; no primer, adhesive, or adhesion-promotion technique needed.
- 2. B Bonding agent needed; candidate systems cross-referenced in Tables 3 through 8, and described in Table 7.
- 3. Blank * Weed for bonding agent highly probable, but not yet determined.

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques

1. Adhesion-Promoting Technique for Low-Soiling Coatings Surface activation with ozone Surface activation with ozone 2. Primer for Bonding EVA to Glass Component Composition Composi	
with the low-soiling costing L-1668 and E-3620; see text experimental description 2. Primer for Bonding EVA to Glass Component Composition Z-6030 silane (Dow Corning) Benzyl dimethyl Amine Lupersol 101 (Pennwalt Corp.) Methanol Primer for Bonding PnBA to Glass Component Composition Comp	rface
Component Cosposition Z-6030 silane 9.0 parts by wt amine Lupersol 101 0.1 parts by wt (Pennwalt Corp.) Methanol 90.0 parts by wt Component Composition Composition Component Composition Composition Component Composition Composition Composition Component Composition Composition Composition Component Composition Composition Composition Composition Composition Composition Composition Frimer developed by Dow Corn for bonding FnBA to glass; pround to give fair performant with Tedlar 100BC30UT plastifilm, and found to be ineffer with metals tested (aluminum copper, mild steel, and sold silicate Isopropanol 180 parts by wt	
Component Composition primer found suitable for bo EMA to glass, and also both and EMA to mild steel (Dow Corning) Benzyl dimethyl 1.0 parts by wt amine Lupersol 101 0.1 parts by wt (Pennwalt Corp.) Methanol 90.0 parts by wt Component Composition found to give fair performan with Tedlar 1008G30UT plastificate [Component Composition with metals tested (aluminum copper, mild steel, and sold silicate Isopropanol 180 parts by wt	
(Dow Corning) Benzyl dimethyl 1.0 parts by wt amine Lupersol 101	nding
amine Lupersol 101	
(Pennwalt Corp.) Methanol 90.0 parts by wt 3. Primer for Bonding PnBA to Class 3. Primer developed by Dow Corn for bonding PnBA to glass; p Component Composition found to give fair performan with Tedlar 1008G30UT plasti Z-6020 silane 10 parts by wt film, and found to be ineffe (Dow Corning) with metals tested (aluminum Ethyl ortho- 10 parts by wt copper, mild steel, and sold silicate Isopropanol 180 parts by wt	
3. Primer for Bonding PnBA to Glass 3. Primer developed by Dow Corn for bonding PnBA to glass; p found to give fair performan with Tedlar 1008G30UT plasti film, and found to be ineffe with metals tested (aluminum Ethyl ortho- 10 parts by wt silicate Isopropanol 180 parts by wt	
Component Composition found to give fair performan with Tedlar 1008G30UT plasti Z-6020 silane 10 parts by wt film, and found to be ineffe (Dow Corning) with metals tested (aluminum thylographics of the film of	
Z-6020 silane 10 parts by wt film, and found to be ineffe with metals tested (aluminum thylographic tested) (aluminum to be ineffe with metals tested (alumi	rimer Ce
Isopropanol 180 parts by wt	ctive ,
and the second s	
 4. Primer for Bonding Z-2591 4. Commercial primer recommende Polyurethane to Glass and Metals Associates; primer is one- 	
Z-3012 component system ready for u	30
5. Primer for Bonding Z-2591 5. Primer developed by Dow Corn Polyurethane to Glass for bonding Z-2591 polyureth glass; primer found suitable	ane to
Component Composition bonding Z-2591 polyurethane Tedlar 100BC30UT and Korad 6	to
Z-6020 silane 5 parts by wt plastic films (Dow Corning) Methanol 95 parts by wt	review E

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques (Cont'd)

	\$y0%.	**		Remarks
Ö 4	Primer for Bonds. Acrylar X-22417 Component 2-6020 (Dow Corning) 2-6030 (Dow Corning) Acroloid AT-51 (Rohm and Haas) Toluene	Composition Sparts by wt I part by wt 100 parts by wt	6 s.	Primer developed by Dow Corning for bonding EVA to Acrylar X-22417 plastic film; as Acrylar is an acrylic film, this primer is evaluated for all material combinations involving acrylic materials; primer has been found suitable for bonding EMA to Acrylar, EVA to Korad 63000 (an acrylic film), and PhBA to Korad 63000; this primer did not work for bonding PhBA to Acrylar X-22417
7*	Adhesive for Bon Tedlar 68040 acryli adhesive	ding EVA to	***	Commercial, one-component, adhesive system available from Du Pont; adhesive also found to work for bonding EVA to Acrylar X-22417 and Korad 63000, for bonding PnBA to Tedlar and Korad 63000, and for bonding a modified EMA (see text) to Tedlar
8.	First Primer for to Tediar Component Z-6030 (Dow Corning) Z-6040 (Dow Corning) Resimene 740* (Monsanto) Methanol *Resimene 714 masubstitute for	Composition 0.25 parts by wt 0.25 parts by wt 10.0 parts by wt 90.0 parts by wt	8.	First primer developed by Dow Corning for bonding EVA to Tedlar that gave satisfactory performance (subsequently an improved version, primer No. 21; was developed); this primer. No. 8, is still included in this table because it was experimentally found to work satisfactorily with ICI Melinex white-pigmented polyester films. Note that primer No. 15 was developed for bonding EVA to polyester films, but did not work with the Melinex polyester film

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques (Cont'd)

***************************************	Syatem	ance), meniphi, is a scrìigh na an imphreochann, a ann did diditing dies criticas disposition in canina i phanesphr	***************************************	Herri K. F. B.
9.	Primers for Bond Polyurethane to		9.	Two experimental primer systems recently developed by Dow Corning; of several exploratory systems
	Component	Camponicion		investigated by Dow Corning, these two yielded equally the highest
	Z-6020 silane (Dow Corning)	9.5 parts by wt		bond strength on freshly made test specimens: however, neither system
	Z-6020 silane (Dow Corning) Water Methanol	0.5 parts by wt 90 parts by wt		has yet been immersed in water maintained at room temperature, or in boiling water; therefore,
	Z-6020 silane (Dow Corning)	5 parts by wt		they are still to be considered experimental
	(Dow Corning) 2-6082 silane (Dow Corning) Methanol	5 parts by wt 180 parts by wt		
10.	Primer for Bondi Polyurethane to Korad 63000		10.	Commercial primer recommended by and available from Development Associates; primer is one-component system ready for use
	2-2881			as received
* * *	Adhesive for Bon Polyurethane to		11.	2-2891 is a white-pigmented polyurethane paint available from Development Associates, formu-
	Z-2891 ureth	ane paint		lated to achieve good adhesion to wood and to 2-2591 polyurethane
12 ×	Adhesive-Primer Bonding Acrylar 63000 to Hardboa Acrylar	and Korad	12.	The 2-2591 polyurethane pottant functions in this application as an adhesive between Acrylar (or Korad) and the hardboard; chemical Sonding of the adhesive to
	Z-2881 Primer (No. 10) Z-2591 Polyurethane Z-2891 Urethane Paint Hardboard	3-Component adhesive and primer System		each of the adherents is accom- plished respectively with the 2-2881 primer, and the Z-2891 urethane paint. All are Develop- ment Associates products. Testing by FSA indicates good initial adhesion, but aging tests not yet done

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques (Cont'd)

	System		Remarks
	Adhesive for Bonding Scotchpar, Acrylar, and Korad 63000 to Hardboard 4910 acrylic pressure— sensitive adhesive		Commercial adhesive marketed by the Adhesives. Coatings, and Sealants Division of the 3M Co., St. Paul, Minn.; the adhesive is dissolved in a solvent, and the adhesive solution can be brushed onto a surface as paint. After solvent drying, the resultant adhesive layer is an instant contact adhesive, and therefore requires care in film alignment. Initial film adhesion is good, but aging tests have not yet been carried out
The second secon	Primer for Bonding EVA to Mild Steel, Copper, and Aluminum Component Composition 2-6030 silane 99 parts by wt (Dow Corning) Zinc chromate 100 parts by wt powder Benzyl I part by wt dimethyl amine Methanol 300 parts by wt	14.	Primer developed by Dow Corning for bonding EVA to mild steel, copper, and aluminum; also found to work in bonding EVA to chrome steel, stainless steel, titanium, and brass. Attempts by different investigators to use this primer for bonding EVA to solder have led to mixed results; some achieve strong bonds, others observe zero bond strength, dry or wet. This necessitated a new primer development for bonding EVA to solder, described in No. 16 below
15.	Primer for Bonding EVA to Scotchpar Component Composition Z-6040 silane 5 parts by wt (Dow Corning) Resimene 740 95 parts by wt (Monsanto) Isopropanol 300 parts by wt	15.	Primer developed by Dow Corning for bonding EVA to Scotchpar polyester film; has been found to work in general with all polyester films except Melinex white-pigmented polyester film, as described in No. 8 above. Also found to work in bonding EVA to Tedlar

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques (Cont'd)

	Syatem		製作物格工製作
	Primer for Bonding EVA to Solder Component Composition 2-6030 (Dow 99 parts by wt Corning) Bensyl I parts by wt Dimethyl Amine Laopropanol 300 parts by wt	The grant of the g	Primer developed by Dow Corning for bonding EVA to solder. Similar to primer No. 14 without zinc chromate, and with substitution of isopropanol for methanol. Presumably the key to successful use is proper and thorough cleaning of the solder surface, and for this, isopropanol is more satisfactory than methanol. Additionally, it is recommended that the solder surface be cleaned before priming with a etone and/or ordinary household kitchen cleanser
	Adhesive and Primer System for Bonding Tedlar to Mild Steel (requires 20 min at 150°C) Tedlar 68040 adhesive EVA (A-9918 3-component formulation) adhesive and Metal primer system (No. 2 or 14) Mild Steel		The EVA A-9918 pottent functions in this application as an adhesive between Tedlar and mild steel; chemical bonding of the EVA adhesive to Tedlar and mild steel is accomplished, respectively, by use of the 68040 adhesive, and one or another of the EVA/metal primers (No. 2 or 14)
18,	Adhesive and Primar System for Bonding Scotchpar to Mild Steel (requires 20 minutes at 150°C) Scotchpar Polyester film primer (No. 15 above) 3-component adhesive and formulation) metal primer (No. 2 or 14) Mild Steel	*** *** ******************************	The EVA A-9918 pottant functions in this application as an adhesive between Scotchpar and mild steel; chemical bonding of the EVA adhesive to Scotchpar and mild steel is accomplished, respectively, by use of the EVA/polyester primer (No. 15), and one or other of the EVA/metal primers (No. 2 or 14)
	Primer for Bonding Z-2591 Polyurethane to Mild Steel I-1022		Commercial primer recommended by and available from Development Associates, Inc.; primer is one-component system ready for use as received

Table 2. Primers, Adhesives, and Adhesion-Promoting Techniques (Cont'd)

	System			Remarks
The second	Adhesive-Primer Bonding Korad 6: Stee! Korad 6:3000 68040 adhesive (No. 7) EVA A-9918 Metal primer (No. 2 or 14) Mild Steel		20.	The EVA A-9918 pottant functions in this application as an adhesive between Korad 63000 and mild steel; chemical bonding of the EVA adhesive to Korad 63000 and mild steel; chemical bonding of the EVA adhesive to Korad 63000 and mild atent is accomplished, respectively, by use of the DuPont adhesive 68040 (No. 7) and one or the other of the EVA-metal primers (No. 2 or 14)
	Second Primer for Tedlar Component 2-6040 silane Dow Corning Cymel 303 (American Cyanamid) Methanol	Composition 1.0 part by wt 9.0 parts by wt 30.0 parts by wt	21.	Second, and better, primer developed by Dow Coring for bonding EVA to Tedlar; also found to work fairly well in bonding EVA to Scotchpar polyester films. Trial testing with bonding EVA to Tedlar and Scotchpar films resulted in very poor performance, but may be explainable by differences in flow properties of EVA versus EMA during processing; see text for expanded discussion
	Primer for Bondi Tedlar Component 2-6032 silane (Dow Corning) Water Acetic acid Methanol	ng PhBA to Composition commenced by wt 5.0 parts by wt 0.2 parts by wt 20.0 parts by wt	· · · · · · · · · · · · · · · · · · ·	Primer developed by Dow Corning for bonding PnBA to Tedlar; also found, in trial testing, to give marginal performance with Scotchpar and Korad plastic files

behavior and no differences in soiling behavior as compared with uncoated controls. This was expected, as FC-721 and FC-723 are not chemically formulated to bond to the surfaces. On the other hand, L-1668 and E-3820 are formulated for chemical bonding to surfaces. Outer covers coated with these two materials have been outdoors for more than 20 months, and some or all of the initially deposited coatings are still on the outer cover surfaces, as evidenced by water bead behavior; soil accumulation remains lower as compared with the uncoated controls. This real-time outdoor behavior indicates at least 20 months of weathering durability for both of the costing materials, and for the interfacial chemical bonds.

All of the outer interfaces can be evaluated by mechanical peel test; the bond-strength values will be found in Tables 3 through 8. These tables are organized by pottants; therefore, Tables 3, 4, and 5 contain bond strengths for EVA bonded systems, Table 6 EMA bonded systems, Table 7 PnBA bonded systems, and Table 8 polyurethane bonded systems.

B. EVA PRIMERS AND ADMESTVES

The first EVA primer system developed was for bonding cured A-9918 EVA to glass (Reference 7). This is primer No. 2 in Table 2, which is also referred to as the glass-EVA primer. Experimental quantities of this primer are available from Springborn, under the designation A-11861 primer. As described in more detail below, this glass-EVA primer resulted in very strong bonding to glass, under both dry and wet exposure conditions. This favorable experience with glass prompted a follow-on evaluation of this primer system for bonding EVA to many other materials (Reference 8). Immediate difficulties, however, were then encountered with copper, aluminum, Tedlar fluorocarbon film and Scotchpar polyester film.

As copper and aluminum represent potential low-cost metals for solar-cell metallization, emphasis was next given to developing a primer system for bonding cured A-9918 EVA to these two metals. The composition of this metal-EVA primer, primer No. 14, is given in Table 3. Another primer system had to be developed separately for bonding cured A-9918 EVA to polyester films such as Scotchpar (3M) and Mylar (Du Pont). The composition of this primer, commonly referred to as the polyester-EVA primer, is also given in Table 2 (No. 15).

For bonding cured A-9918 EVA to Tedlar film products, Du Pont identified an acrylic contact adhesive designated 68040 (No. 7, Table 2). As supplied by Du Pont, the acrylic adhesive system is a solution in toluene, which is spread on the Tedlar surface and allowed to dry. The resultant adhesive coating is dry and nontacky, and the coated Tedlar film can be wound and unwound. Experimental work with this adhesive system has used a costing thickness of 0.3 to 0.4 mil, which has yielded acceptable performance. The contact adhesive develops its bonding qualities at the high temperature of the EVA lamination cycle. A PV manufacturer may coat Tedlar optionally with the adhesive as part of its manufacturing operation, or arrange to have the coating put on by an independent coating vendor.

Table 3. Adhesive Bond Strengths for EVA (A-9918) Bonded to Various Materials

		Bond Strengths, lb/in. of width								
Materias	Primer Number	Control	6 days	2 wk Immersion	2 hours Boiling Water					
Sunader glass	2	34.8	- New .	30.0	92.3					
Window glass	2	39.6	***	37.9	27					
Window glass (self-priming EVA)	2	35.4	.4004	41.9	Cohesive					
Calvanized steel	iraninga ngangangangangangangangangangangangangan	2.3	440							
Mild steel	2	56.0	Water	42.6	50.7					
Mild steel	14	400	17.6	wax	inei					
Mild steel	21	2.3	3884	2.3	3.9					
Aluminum	2.	41.0		2.3	2,.6					
Aluminum	14	**	Cohesive	Constant of the Constant of th	**					
redlar 1008630Ut	2	4.5	***	0	0					
Tediar 100BG30UT	Ÿ	6.2	***	6.5	7.8					
rediar 100BC30UT	8	19.0	:400	6.3	6.5					
Tedlar 100BG30UT	21	30.0	Seeds,	12.6	13.0					
scotchpar 20CP WII	15	35.3		31,3	21.3					
Scotchpar 20CP WH	21	11.5	××.	26,8	1 × 1.					
Acrylar X-22417	2	1.0		Ö	<u> </u>					
Acrylar X-22417	6	15.4	** *:	'wa:	****					
Acrylar X-22417	7	Cohesive	:### ;	Cohesive	Cohesiv					

Table 3. Adhesive Bond Strengths for EVA (A-9918) Bonded to Various Materials (Cont'd)

Materials		Bond Strengths, lb/in. of width			
	Primer Number	Control	6 days Immeration	2 wk Immereion	2 hours Boiling Water
Plexiglas V-811		5,6	***	7.8	5.5
Plexiglas V-811	7	40.2	-2019 v	444	44.1
Copper	14	32.5	Cohesive	Cohesive	37.6
Copper	16	5.1	***	40.7	3.9
Korad 63000	6	Cohesive		Cohesive	Veak
Korad 63000	*	Cohesive	***	5.4	Cohesiv
Korad 63000	21	1.0	\$600.	- in	-3002
Korad 212	2		yan	**************************************	
Tedlar 150BL30WH	7	Conesive		Cohesive	Cohesiv
Titanium	14		Cohesive	ing .	i
Brass	14	488	Cohesive	wij	iii
Stainless steel	14	***	19.8	in	***
Chrome steel	14	****	Cohesive	eia:	440,
Nickel	2	***************************************	in in	Test	an in a single property in the contract of the

^{*}Similar chemiatry as Acrylar films, but available in thicker versions in order to prepare mechanical peel-test specimens. Testing Acrylar peel specimens is experimentally difficult due to film's low fold resistance.

Table 4. Adhesion of EVA (A-9918) to Solder Using Primer No. 16, Showing Importance of Surface Cleanliness for Solder Bonding

Bond Strengths: 16/in. of width Solder Surface Method of After I Day Condition Cleaning Primer Jeed in Water Dry. Dall 0 8.8 None No 04.11 Mone Yes 15.,4 11.0 Dull Acetone Yea Cohesive Cohesive Dull Cleanser Cohesive Cohesive Yes 0 Shiny None No 6.6 Ō. Shiny None Yes 8.8 Shiny Acetone Yes Cohesive Cohesive Shiny Cleanser Yes. Cohesive Cohesive

Table 5. Adhesion of New*, Experimental A-15295 EVA to Sunadex Glass and Mild Steel

Bond Strengths, 15/in, of width

magnituren en oardaministatiin jojanna karimen ja monto tataman jointiin en oa mantamatan jointiin ja monto ma Taringan en on				
Primer	Control	2 wk Water Immersion	2 hours Boiling Water	
**	51.3	32.9	33.3	
	15.6	3600	Same.	
	Number 2	Primer Control 2 51.3	Primer 2 vk Water Number Control Immersion 2 51.3 32.9	

^{*}This EVA incorporates an experimental peroxide curing agent, designated TBEC, in substitution for Lupersol 101 (used in A-9918 EVA). Other than this substitution, A-15295 EVA is the same as A-9918 EVA.

Table 6. Adhesive Bond Strengths for EMA Bonded to Various Materials

Bond Strangths, 1b/in. of width YMA Primer or 2 hours Peroxide Boiling Adhesive Curing 2 wk Water Materials Immersion Number Control Water Agent 40.0 Sunadex glass 1.-101 60.2 27.7 64.2 45.5 Sunadex glass L-TBEC Glass broke Mild steel 2 1-101 5.,6 14.6 13..9 Mild ateal 2 L-TBEC 35.1 2.8 Low Tedlar IOOBG30UT 7 L-101 0.5 NT MT Tedlar 100BG30UT 0.8 21 1-101 NT MT Tedlar 100BG30UT 7 L-TBEC Cohesive MT Cohesive 0.4 Scotchpar 20CP WH 21 1,-101 0.8 1.2 Tedlar 150BL30WH 7 1-101 1.9 NT NT Acrylan X-22417 6 L-TBEC Cohesive Cohesive Cohesive Plexiglas V-811 3.6 6.4 L-TBEC LOW

Table 7. Adhesive Bond Strengths for PnBA (BA-13870) Bonded to arious Materials

Materials		Bond Strengths, lb/in. of width			
	Primer or Adhesive Number	CASE & E & I	2 wk Water Immersion	2 hours Boiling Water	
Sunadex glass	20026	arani orani orani and	n Orani in in ini in antana in manana manana na ma Manana na manana na m	optitutitututus aastataa kooko kooko kooko jo mijoriisto een asaa aasta kooko kooko kooko kooko kooko kooko ko eesta	
Sunadex glass	3	3.0	1.2	1.4	
Sunadex glass	22	0.9	0.4	0.6	
Tedlar 1008G30UT	in and the second s	<u> </u>	ine-inergia isteriologica que constitutativa de la infrience en enque ine-	n 1986 in	
Tedlar 100BG30UT	.3	1.3	PnBA failure	0.8	
Tedlar 100BG30UT	où Y	PnBA failure	PnBA failure	PoBA failure	
Tedlar 100BG30UT	22	2.4	2.4	23	
Scotchpar 20CP WH	None	Ö	iking gina animan miningan animan menungan nganggan "Man	ana ta'u ta'u ta'u ta'u ta'u ta'u ta'u ta'	
Scotchpar 20CP WH	22	1.4	% ***	Soundaire.	
Korad 63000	None	0			
Korad 63000	*	8.3	YnBA failure	3.4	
Korad 63000	7	PnBA failure	PoBA failure	PnBA failure	
Korad 63000	22	0.7	AMI -	- Naile	
Acrylar X-22417	\$00 8	<u> </u>	. 1999.	dar	
Acrylar X-22417	6.	Lów	- industrial control of the control	- 46h*	
Acrylar X-22417	7	Very low	Very low	'ogc	
Aluminum	aneste et et en inicialization de la companie de l La companie de la co	C.	. Mar		
Copper	3.	Ö.	° èssè-	**************************************	
Mild steel	3	Ŏ.		1999	
Solder	3	Ô	in.	**	

Table B. Adhesive Bond Strengths of Z-2591 Polyurethane to Various Materials

Bond Strengths, 15/in. of width

Materials	Primer or Adhesive Number	Control	2 wk Water Immersion	2 hours Boiling Water	
Sunadex glass	Norte	0	andro or de serigi esta de la passa de la maine a regis e espacio espacio de mediando menso de la virgida de s Asserta	internation in the second control of the sec	
Sunadex glass	n ii	31.4	37.2	45.2	
Tedlar 100BG30UT	None	<u> </u>	e distribution de la companya de la Companya de la companya de la compa	anna kana kana makana makana kana kana k	
Tedlar 100mc30UT	5	4.8	1.4	13	
Tedlar 1008G30UT	94; 98	7.0	èlesi	soper	
Scotchpar 2002 VII	Nexte	0	et english dan latan milimini miliyin tan english da umama mengapisa angapish dalah da 1866	ina	
Scotchpar 20CP WN	5	0.2		. Acroide	
Scotchpar 200P WH	.8:	low	iov.	100	
Korad 63000	New 13 lb		in en	anningigy, i deriving kanagarian er er engliken. 	
Korad 63000	\$	3.6	2.6	2.7	

Note: The commercial primers available from Development Associates, No. 4, 10, 11 and 19, have not yet been evaluated by FSA.

Table 3 is a master table of adhesive bond-strength measurement accumulated for cured EVA bonded to a variety of materials, with various combinations of primers and the 68040 adhesive. These data, generated at different experimental laboratories, result from a similar experimental pattern: measurement of control values at ambient conditions, followed by measurement (again at ambient conditions) of specimens that have been immersed in water maintained at room temperature, and of a ecimens immersed in boiling water. The exception is work carried out with the metal-EVA primer system, No. 14, for all metals but copper. Bond-strength measurements for these test specimens were made after 6 days immersion in water maintained at room temperature, and no control values were measured.

The glass-EVA primer (No. 2) can be used optionally as either a wipe-on primer or as a compounding additive to generate a self-priming EVA. If used as a compounding additive, the methanol solvent is not used, and the three-component mixture is blended into the EVA at a concentration of 1.0 wt %. A concept used in the formulation of this primer is that the addition of a small amount of the Lupersol 101 peroxide causes a localized generation of active free radicals during the heating and curing of the A-9918 EVA, resulting in a substantially higher crosslinking at the polymer-substrate interface.

The glass-EVA primer was tested by priming clean soda-lime glass slides with a thin layer of the primer (wiped on with a moistened pad) and air drying for 15 min. The fully formulated EVA compound (A-9918) was then compressionmolded and cured against the surface. The resulting specimen was tested for peel strength by ASTM D-903. The results (Table 3) were excellent, yielding an average peel strength of 39.6 lb/in. of width. Duplicate specimens placed in boiling water for 2 hours and evaluated by the same process yielded an average peel strength of 27 lb/in. of width, still excellent adhesion. A quantity of the primer formulation was prepared without the alcohol diluent and blended into the standard A-9918 EVA formulation at 1 wt % to test the self-priming effect. The resulting resin was then compression-molded and cured against a clean glass slide, as before, and tested for peel strength by the same method. The average strength was 35.4 lb/in. of width (Table 4). The bond strength of the self-priming EVA after room-temperature water immersion for 2 weeks increased from the initial value of 35.4 to 41.9 1b/in. of width, and resulted in cohesive failure after 2 hours in boiling water.

As shown in Table 3, the glass-EVA primer (No. 2) not only gives excellent and durable bonds to regular soda-lime (microscope slides) and low-iron (Sunadex) glass, but also is marginally effective with aluminum, and very effective with mild steel. Aluminum, primed with A-11861 (primer No. 2) gave an average dry-bond peel strength of 41.0 lb/in. of width and the adhesion to mild steel was even higher, with an average dry strength of 56.0 lb/in. of width. This is the highest strength found between EVA and any other material. The wet-bond strength is excellent to mild steel and poor to aluminum. Dry-bond strengths to Tedlar were very low, and virtually zero when wet. The A-11861 primer was essentially ineffective for copper and for Scotchpar polyester film, for both dry and wet exposure conditions. On the other hand, the polyester-EVA primer No. 15 worked excellently, yielding a dry control value of 35.3 lb/in., dropping slightly to 31.3 lb/in. after 2 weeks of room-temperature water immersion, and still retained a high value of 21.3 lb/in. after 2 hours of immersion in boiling water.

Metal-EVA primer No. 14 achieved outstanding bonding of cured A-9918 EVA to aluminum and copper (Table 3), with peel testing resulting in cohesive failure of the EVA after 6 days of water immersion. Cohesive failure of the cured EVA also occurred for bonding to chrome steel, titanium, and brass. The metal-EVA primer is apparently not as effective with mild steel as the glass-EVA primer, A-11861, but the bond strength values of 17.6 lb/in. is still acceptably high.

The two Tedlar films listed in Table 3, Tedlar 100BG30UT and Tedlar 200PT, are clear and transparent UV-screening films. The former is 1 mil thick and the latter is 2 mils thick. The third Tedlar film listed in Table 3, Tedlar 105BL30WH, is a 1.5 mil-thick white-pigmented film. The bond strength data given in Table 3 reveal that the 68040 adhesive yielded comparable wet and dry performance for the two UV screening films, and yielded outstanding performance with the white-pigmented Tedlar film. Involvement of the white-pigment material in enhancing the bond strength can only be conjectured.

With all three Tedlar films, the adhesive 68040 retained wet-bond strength, as compared with the zero bond strength of the glass-EVA primer No. 2. Recently Dow Corning Corp. developed two additional, still experimental, primer systems for bonding Tedlar films to EVA. These primers are designated as primers No. 8 and No. 21. With primer No. 21, control bond strengths were near 30 lb/in. After 2 weeks of water immersion the bond strength dropped to 12.6 lb/in, and 2 hours of exposure in boiling water reduced the bond strength to near 13 lb/in.

One of the low-cost structural panel materials being investigated for module application is mild steel (p. 3), specifically cold-rolled mild steel. However, the concern with this material in an outdoor application is corrosion. There is literature evidence (References 12 through 15) that indicates that use of chemical bonding agents can prevent metallic corrosion. This approach to corrosion prevention in mild steel is being evaluated experimentally by using a chemical bonding adhesive system for bonding white-pigmented plastic films to both surfaces of mild-steel panel. A trial film-adhesive-primer system has evolved (No. 18), which consists of the following layers of materials on mild steel:

Outer film Scotchpar 20CP WH

Primer Polyester primer No. 15

Adhesive Cured A-9918 EVA

Primer Metal primer No. 14

Panel Mild steel

Panels of mild steel coated with this film-adhesive-primer system have survived more than 4,500 hours of continous exposure to salt spray (ASTM B-117) without any evidence of corrosion (Reference 10). Unprotected mild steel, on the other hand, begins to corrode within hours after exposure to salt spray. This use of EVA as an adhesive offers an another possible application area for these low-cost materials in terrestrial photovoltaic modules.

Very recently, laboratory attempts to bond EVA to solder with metal. primer No. 14 were unsuccessful, usually with little to zero dry bond strength; this prompted a directed effort focused on EVA-solder bonding, which resulted in primer No. 16. It should be noted that this is essentially primer No. 14 without the zinc chromate, and that isopropanol solvent has been substituted for methanol. The substitution of isopropanol, which is a better surfacecleaning agent than methanol, emphasized the important requirement of surface cleanliness for EVA-solder bonding. This is shown in the bond-strength data in Table 4, where precleaning of the solder surface either with acetone or with a common household kitchen cleanser resulted in cohesive bonding of EVA to solder. This work was performed with a variety of solder specimens whose surfaces could be qualitatively characterized as shiny or dull. No effort was made to find an explanation for the differences, even though it was noticed that a primed, dull surface without pre-cleaning gave better dry and wet performance than did a primed, shiny surface that also was not precleaned. In each case, however, precleaning appeared to eliminate any bonding effects related to initial surface quality.

Since it is desired to have a single metal primer for EVA rather than two, a study is now under way that will attempt to identify a single primer combining the features of primers No. 14 and No. 16, and the cleaning procedures described above.

Recently (Reference 8), Springborn identified a new peroxide curing agent for EVA, which results in a faster cure at lower temperatures. The new peroxide is marketed by the Lucidol Division of Pennwalt under the designation Lupersol-TBEC, or L-TBEC for short. The standard peroxide currently employed in A-9918 EVA is marketed by Lucidol under the designation L-101. The potential substitution of L-TBEC for L-101 in advanced EVA formulations prompted concern for its possible effects on EVA adhesion. Two preliminary adhesion test results for Sunadex glass and mild steel with TBEC-cured EVA are shown in Table 5. The results suggest no problems with adhesion to Sunadex glass, although interestingly there is an unexplained increase in the dry (control) strength as compared to L-101-cured EVA. The wet strengths are essentially the same. The dry (control) strength with mild steel is considerably reduced (from 56.0 to 15.6) for the TBEC-cured-EVA, as compared with the L-101-cured EVA, but any conclusions or judgments at this time would be premature. The wet strengths were not available at the time of this publication.

C. EMA PRIMERS AND ADHESIVES

In terms of peroxide curing and chemical bonding, EMA is chemically very similar to EVA. Therefore the first curing formulation for EMA was based on L-101 chemistry, and worked well for EMA (Reference 11). Subsequently, with the identification of L-TBEC, this was evaluated for EMA and found to result in a faster cure at lower temperatures (Reference 8), as observed with EVA. Two versions of EMA are now under evaluation, one with L-101 and another with L-TBEC. Adhesion results accumulated for both versions are reported herein.

Because of chemical similarity, primers and adhesives that have been found to work for EVA become trial candidates for EMA. Table 6 summarizes the results. The first primer effort was with the primer developed for EVA-glass

(No. 2), which yielded for glass and L-101-cured EMA a very high control value of 60.2 lbs/in., very respectable wet-bond strength values of 40.0 lb/in. after 2 weeks of water immersion, and 27.7 lb/in. after 2 hours in boiling water. Actually, the wet-bond strength values for L-101-cured EMA-glass are almost the same as those for L-101-cured EVA-glass (Table 3), with the major difference being the dry control values. The adhesion to glass with TBEC-cured EMA is remarkable. The glass breaks during efforts to measure the dry bond strength mechanically, and wet strengths are extremely high.

With the two exceptions of TBEC-cured EMA bonded to Acrylar and Tedlar, using EVA primer and adhesive systems, the general trend is toward lower control and wet-bond-strength values, as compared with bonding of these same materials to EVA. There are, however, indications from the mild steel and Scotchpar test results that wet-bond strengths with L-101-cured EMA tend to be higher than the initial control values, indicating that time, water, and/or heat may be promoting additional bonding reactions, suggesting strongly a possibility that full bond-strength potential was for some reason not initially developed.

One explanation may be related to differences in the high-temperature processing characteristics of EVA and EMA. Above 70°C, EVA becomes a viscous fluid that will facilitate spread and wetting over a mating surface, and therefore will tend to increase the intimacy of contact and interpenetration with a coupling agent deposited from a primer solution. On the other hand, EMA does not become fluid upon heating; it becomes a softer solid that requires some pressure for flow and spread over a mating surface. Thus intimacy of contact and interpenetration with a deposited coupling agent may be reduced, as compared with EVA. Since the data trends in Table 6 indicate that primers and adhesives found for EVA appear to work with EMA, perhaps physical effects such as the one just described, rather than chemical effects, are initially limiting achievements of full bond-strength potential. This area of EMA bonding will be investigated. There is also a hint in the data of Table 6 that L-TBEC peroxide seems to enhance the development of higher interfacial bond strengths; this also will be investigated.

EMA, like EVA, is a dry laminaton film at room temperature, and is formulated to undergo peroxide curing to a thermosetting elastomer at processing temperatures in the order of 150°C. For EVA, peroxide crosslinking is an absolute necessity, in order to generate an encapsulation pottant having thermal stability against fluid flow at service temperatures above 70°C. Accordingly (and this is very important), all of the primers developed at Dow Corning for EVA are formulated to generate their interfacial chemical bonding as an integral part of the high-temperature EVA crosslinking reaction. Since the same crosslinking formulation is compounded into EMA, the Dow Corning primers developed for EVA will require that EMA undergo the peroxide crosslinking reaction.

However, the need for EMA to undergo peroxide crosslinking is not clear, as it is for EVA, and some preliminary module testing using uncured EMA as a pottant suggests that it has adequate thermal stability against flow and creep at temperatures up to $90^{\circ}\mathrm{C}$. The elimination of a potentially unnecessary processing step for EMA is attractive, but the chemical requirements for EMA bonding agents would then be changed. Indeed, efforts to use primer No. 2 for

non-curing EMA against glass were unsuccessful, with no wet strength, and little to zero dry strength.

New work to develop bonding agents specifically for non-curing EMA is under way. The first candidate for non-curing EMA on glass is as follows:

Component		Congr	1011	100
Z-6020 silane (Dow Corning)	2	parts	by	weight
Parlon-5CS (Hercules Inc.)	100	parts	by.	veight
Toluene	100	parts	by	veight

The control bond strength for non-curing EMA-glass using this primer was near 22 lb/in, and after 1 day in room temperature water was 11.7 lb/in. These initial results are encouraging.

D. PABA PRIMERS AND ADMESIVES

Poly-n-butyl acrylate (PnBA) is a liquid casting acrylic that cures to a temperature-stable, clear white elastomer (Reference il). Specifically, the PnBA in liquid form is a polymer-monomer syrup, which is a solution of poly-n-butyl acrylate in n-butyl acrylate monomer. Included in the syrup mixture are additives that activate, at elevated temperatures, the polymerization of the n-butyl acrylate monomer, and then subsequent crosslinking of the total polymer system.

Early attempts to develop bonding agents for PnBA quickly revealed that the n-butyl acrylate monomer was functioning as a solvent for the deposited coupling agents on a mating surface. The n-butyl acrylate would apparently dissolve away most of the coupling agent, leaving the interface with significantly reduced capacity for any substantial interfacial chemical bonding.

Chemical efforts to counter the dissolution problem have resulted in the development of two primer systems intended specifically for PnBA, primers No. 3 and 22; the performance results of these primers are given in Table 7, along with some results using primer No. 6 and adhesive No. 7. It should also be pointed out that PnBA apparently has a very low cohesive strength, somewhere in the range of 3 to 8 lb/in. This naturally low cohesive strength, plus the solvent action for coupling sgents, creates a difficult task in assessing the true level of interfacial bond strength by mechanical peel testing. PnBA systems are prime candidates for study of their chemically bonded interfaces by the chemical hnique (Reference 9) discussed in Section III. The development of bonding agents for PnBA is a major program effort.

E. POLYURETHANE (2-2591) PRIMERS AND ADRESIVES

Z-2591 is a liquid-casting, clear white polyurethane (PU) system available from Development Associates. Development Associates markets a variety of proprietary primers for this polyurethane, each appropriate to the

material that is to be bonded to the polyurethene. Primers No. 4, 10, 11, and 19, available from Development Associates, were recently recommended by them for glass and metals, Acrylar and Korad 63000, hardboard, and mild steel respectively. These primers have not yet been evaluated by FSA.

Dow Corning had developed primer systems for many industrial polygrethene materials, and was able to suggest primer systems for evaluation from their industrial experience. These are primers No. 5, 9(a), and 9(b); results of performance testing with these primers are given in Table 8. Primer No. 8 was also tested with Scotchpar. It should be noted that primer No. 5, the PU-glass primer, performs very well and almost identically with primer No. 2 for both EVA and EMA with glass. Primer No. 5 worked acceptably with Korad and Tedlar. The other primer systems, 9(a) and 9(b), yielded a higher control value with Tedlar than did primer No. 5, but wet values have yet to be taken. Further FSA work on PU bonding agents will include the testing of the primers recommended by Development Associates.

F. ADDITIONAL NOTES

In the introduction to this section it was remarked that long-term aging tests of the bonding agents has yet to begin. This is only partially true, as one coupling agent (2-6020, Dow Corning), and one adhesive system (No. 18) have been aged as a consequence of testing for other purposes.

Z-6020 is the coupling agent used to bond perfluorodecanoic acid chemically to outer-cover surfaces, to generate a low-soiling surface coating (designated E-3820 on p. 2 and in Table 1). The E-3820 coating has been outdoors for more than 20 months, being directly exposed on surfaces to the natural weathering elements consisting of UV, oxygen, moisture, dirt, and other atmospheric chemicals and pollutants. The low-soiling property is still functional at 20 months, suggesting the continuing presence of the surface coating, and thereby implying the continuing presence of the Z-6020 coupling agent on the surface. If so, weathering stability of the Z-6020 is also implied; this holds some promise of weather stability for all of the primers containing Z-6020 listed in Table 2.

Adhesive system No. 18 is used to bond Scotchpar polyester film to mild steel adhesively, for evaluation as a corrosion protection coating. Panels of mild steel coated with Scotchpar and No. 18 adhesive system have survived more than 4500 hours of exposure to ASTM B-117 salt spray (which also exposes the test speciments to 100% relative humidity at 100°C), and to a continuing condensation of hot water. This harsh environment caused no corrosion of the protected mild steel, and also did not affect the protection system, which included two coupling agents, Z-6030 and Z-6040. The stability of these two coupling agents in atmospheric oxygen and humidity at high temperatures holds promise of stability for all of the primers containing Z-6030 and/or Z-6040 listed in Table 2.

THEORIES OF THE CHEMICALLY BONDED INTERPACE

A. INTRODUCTION TO INTERPACIAL BONDING THEORY

To achieve 20- to 30-year life, the various components of solar-cell modules should be held together by strong adhesive bonds, with the interfacial bonding materials being durable and resistant to delamination, corrosion or other failures resulting from the weathering environment. The scope of potential adhesion problems associated with current and future module designs is very broad because of the widely diverse nature of the component materials (glasses, metals, organic polymers, wood, etc.). While a unifying theory of adhesion and performance is not yet achieved, fundamental principles governing interfacial bond formation and durability common to many material combinations are emerging. A large portion of the adhesion research activities associated with solar modules has involved bonding of organic polymers to inorganic substrates (glasses, metals). Consequently, the discussion of interfacial bonding theory that follows draws heavily on results from many sources active in this aspect of bonding technology for demonstration of fundamental concepts.

The work of adhesion, W_A , which is the measure of the strength of a bond between two surfaces, is proportional to their area of contact. Many solid adherends do not have surfaces that will conform to each other on contact (Figure Ja). The gaps between two such solids are filled with air or contaminants, which usually lower W_A . An adhesive is simply a gap-filling material placed between the two solid adherends to establish the intimate molecular contact at the interface that is necessary for developing strong joints (Figure Jb). Interfacial adhesion may be achieved by any of several means, including:

- (1) The action of attractive forces that are physical in nature.
- (2) The formation of primary (covalent or ionic) chemnical bonds bridging the adhesive-adherend interface.
- (3) Molecular penetration (interdiffusion) of the adhesive-adherend molecules.

In most instances, bonding results from a combination of these mechanisms.

The importance of establishing strong, durable interfacial bonds to the integrity of solar modules can be inferred from composites technology, where it is known that degradation of the material as a whole is dominated by degradation of the interfacial bonds between the matrix and the reinforcing fibers. If any of the interfacial bonds in solar modules fails, the performance of the entire module may be jeopardized. There are three principal modes of interfacial bond failure:

- (1) Physical displacement of the interface.
- (2) Debonding at the interface due to chemical reactions.
- (3) Cohesive failure of adhesives or adherends.



The failure mode of a particular bond (if indeed it fails) will be determined by the environmental stresses, including moisture, temperature, UV light, and mechanical loading to which it is subjected.

Studies of bonding in solar modules have emphasized water as the primary environmental agent of concern with regard to bond durability. It is found that the kinetics of failure for unprotected adhesive joints often are governed by diffusion of water into the joint. To increase the moisture durability of solar-module interfacial bonds, coupling agents or adhesion promoters have been widely used. Organofunctional silanes have received the most attention in this connection because of their proven moisture resistance and ability to bond to a wide variety of materials, including metals, metal oxides, glasses and organic polymers.

As has been pointed out in preceding sections of this report, very little data are available on the aging of bonded interfaces. More work is required to assess the potential for failure due to long-term exposure to water and other environmental stresses.

B. PRINCIPLES OF PHYSICAL ADMESION

When any materials are brought into close contact with each other, forces act between the atoms on their surfaces that are totally independent of any chemical interactions. The nature and strength of these forces control all non-chemical surface phenomena, including wettability and physical adhesion.

Material situated on the surface of a body is at a higher energy than that in the bulk. Atoms situated below the surface experience attractive and repulsive interactions with their nearest neighbors in all directions. These interactions are responsible for the cohesive strength of the material. Atoms on the surface however, can interact with only half as many nearest neighbors, which are in or below the plane of the surface. This situation gives rise to physical adhesion. When the surfaces of an adherend and an adhesive are brought into intimate contact, the atoms on one surface interact with those on the other in a manner similar to that of the nearest-neighbor interactions in the bulk of a homogeneous body. Physical adhesion occurs if there is a net attractive force acting across the interface between the atoms on the two surfaces.

Regardless of bonding ability, a strong interfacial bond will not result unless the adhesive spreads spontaneously over the adherend surface, leaving no air gaps when the joint is formed. Consequently, before considering the potential strength of a bond, it is necessary to ensure wettability of the adherend surface by the adhesive. A criterion for wettability is that the adhesive solution exhibit zero (or near zero) contact angle on the adherend surface at some time during the bonding process. The contact angle θ , an experimentally measured parameter of a liquid-solid interface, is defined as the angle between the plane of the solid surface and a line drawn tangential to the surface of a small liquid droplet at the point where it contacts the surface (Figure 5). It is easy to see that θ going to zero corresponds to the collapse of the droplet as it spreads across the surface.

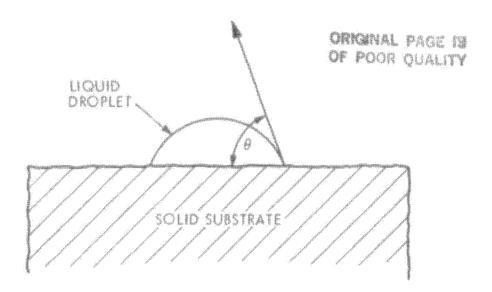


Figure 5. Diagram of a Liquid Droplet on a Solid Surface, Showing the Contact Angle (the Angle Between the Solid Surface and a Line Drawn Tangential to the Droplet's Surface at the Point Where it Touches the Solid Surface)

The contact angle, although a fundamental measure of wettability, is not a practical criterion in module design because it is a property of the combination of adhesive and adherend. For material selection purposes it is desirable that wettability be predictable from inherent properties of the individual materials rather than their combination. Fortunately, that is possible. Figure 6 shows a plot of $\cos\theta$ vs $\gamma_1 v$, the surface tension, for a number of liquids on two polymer surfaces (Reference 16). That value of $\gamma_1 v$ where $\cos\theta$ reaches unity is called the critical value of the surface tension and represents the point where θ goes to zero. From a design point of view, a more suitable criterion for ensuring spreading of an adhesive on an adherend's surface is then that the surface tension of the adhesive solution must be less than the critical surface tension of the adherend surface, or

adhesive adherend
$$\gamma_{\rm LV} \leq \gamma_{\rm Crit}$$
 (1)

Values of these parameters are available in the literature or from manufacturers for a variety of materials.

If the surface of an adherend is suitably wettable by an adhesive, then it is possible for a bond to be formed. The bond strength is typically described in terms of the work of adhesion, which is the energy required per unit bonded area to separate the adhesive bond. The greater the value of $W_{\rm A}$ as measured by, for example, a peel test, the stronger the bond.

The inte atomic forces responsible for physical adhesion are usually divided into two classes, polar and dispersion. The contributions of each type of force can be determined from surface free-energy measurements. An expression for the work of adhesion can then be written in terms of the

ORIGINAL PAGE 18 OF POOR QUALITY

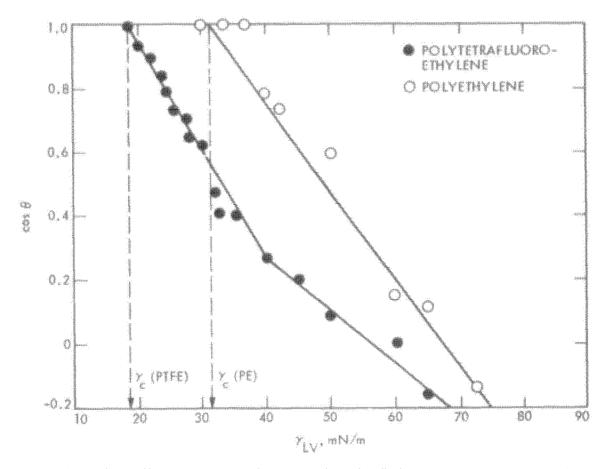


Figure 6. Plot of $\cos\theta$ vs $\gamma_{\rm LV}$ (Surface Tension) for a Number of Liquids on the Surfaces of Polytetrafluoroethylene and Polyethylene

polar-force and dispersion-force components of the surface free energies, $\gamma_{\rm p}$ and $\gamma_{\rm D}$ respectively, of the adhesive and the adherend. The work required to separate a joint formed between materials designated A and B (Reference 16) is:

$$W_{A}^{AB} = W_{P}^{AB} + W_{D}^{AB} = 2 \left[\left(y_{P}^{A} y_{P}^{B} \right)^{1/2} + \left(y_{D}^{A} y_{D}^{B} \right)^{1/2} \right]$$
 (2)

Basically, Equation 2 says that polar materials tend to bond well to other polar materials, and materials with large dispersion forces tend to bond well to other materials with large dispersion forces; in other words, like bonds to like.

Silane primers are used extensively in the production of solar modules to bond to organic and inorganic substrates. While the chemical bonding ability of these primers is usually emphasized, they also exhibit physical adhesion properties. Table 9 lists experimentally measured values of $\gamma_{\rm p}$ and $\gamma_{\rm p}$ for a series of commercial silanes. It is seen that through chemical substitution, the properties of these materials can be altered so as to bond physically to

Table 9. Surface Energy Dispersion and Polar Force Components of Polymerized Silane Coupling Agents With Structure R-Si(OCH3)3

%-Structure	Commercial Designation	⁷ 0 4yn/cm	Zy Zyn/ca
H ₂ N(CH ₂) ₂ NH(CH ₂) ₃ *	06 2 2 1 6 0 2 0	anaaniteennisis seemuunin kan kan kun kun kun kun kun kun kun kun kun ku	4 . C
CH ₂ Q CH ₂ WC~C~O~(CH ₂)3**	ola punita tau unum minyempen ari	alle aurorinine et en et en et en et en et en	inananananinining orangang
CH2*CH*CH2*O*(CH2)3* (Catalyzed)	aasimuutuseta kin saa aasimatoo ka aalaataa kin oo	noisimus interiores in electricales in interiores in interiores in interiores in interiores in interiores in i 1,0 , 2	tanananananin'i menengehinya sinjangi sanjangi sanjangi sanjangi sanjangi sanjangi sanjangi sanjangi sanjangi A 3 4 6
$\operatorname{CH}_2^{\overset{\circ}{\operatorname{CH-CH}}_2-\operatorname{O-(CH}_2)_3}$ (noncatalyzed)	where $z \sim 500$	1.7.46	2.5.44
alata (CH y) z ur	DC XZ-8-0999	idan menengan ngangan katangga kapatangga katangga kapatangga kapatangga kapatangga kapatangga kapatangga kap Si Si S	3 x E
nopurationis muumus en opia suurinny sissä saaninny sissä saaninnuus muusia saanin siin ole ole osaa. Nilly mii (Olly sai) y	ami merrepassi ki esimpo peripa kamini merrepa pengangan pengangan pengangan pengangan pengangan pengangan pen UC A-1100	moinneissen, seisseissennaataisimintemmeeneenassa	19.8
isansi saatiin kanamaan oo ay	000 XZ-8=0951	isa innum sequenceperagues esperagues innivirus especa. E 2 4 4	, O , O
oo oo oo aanaa aanaa aa aa aa aa aa aa aa aa aa	dansunimmente insus singrison ministeri e dascidanda asierem memeretagan nya kangasasian atau si		

almost any desired substrate. Dow-Corning's Z-6030, for example, would be expected to bond well to highly polar surfaces while Z-6020 should bond well to materials with a large dispersion-force component.

Equation 2 works well for predicting bonding ability of organic polymers, inorganic glasses and similar materials. However, in many cases, bonding to metal surfaces is described in terms of a different formalism. This results from the fact that many adhesion properties of metals are dominated by the acid-base interactions of the oxides that cover metals, under ambient atmospheric exposure (Reference 17). As pointed out by Parks (Reference 18) and Bolger and Michaels (Reference 19), experience shows that different metals and their oxides display widely different types of interactions with adhesives and corrosion agents based upon the isoelectric point of their surface (IEPS).

The IEPS is defined as the pH where the immersed metal oxide surface has zero net surface charge or where the concentration of hydrogen and hydroxide ions on the surface are equal. The IEPSs of common metal oxide surfaces in water are given in Table 10 in order of increasing basicity.

The principles of bonding polymers to metal surfaces are, however, still not well known. The adhesion of polystyrene and functionalized styrene copolymers to various metal surfaces has been studied (Reference 7) and no consistent pattern of performance is discernible. For example, acid-modified copolymers do not necessarily form the strongest, most durable bonds with alkaline (heavy metal) surfaces, nor are amine-functionalized materials necessarily best on acid surfaces. Clearly, this is an area that requires further study.

C. CHEMICALLY BONDED INTERFACES

In the laboratory, W_A is always meass of under dry, inert conditions. In the field, however, solar modules are subjected to environmental attack. Experience indicates that under wet conditions or exposure to moist atmospheres at high temperatures, the strength of bonded interfaces decays with a rate influenced by stress, temperature and relative humidity.

In general, an adhesive bond should not be pictured as a static sandwich of an interface region between adhesive and adherend, but as a dynamic equilibrium of an adhesive competing with various potential-weak boundary layers of the adherend surface. In polymer-to-organic adhesion, the weak boundary layers will most likely be oils, low polymers or other additives of

Table 10. Isoelectric Points of Metal (Oxide) Surfaces in Water

Oxide Surface	XXXX	Oxide Surface	IEPS
Silicon	2.2	Aluminum	9 , 1
Manganese	4.2	Zinc	9.40
X in	4.5	Copper	9,1
Titanium	6.0	Nickel	11.0
Zirconium	6.5	Iron (reduced)	12.0
Chronium	7.0	Magnesium	12.2
Iron (oxidized)	8.5	Siver	> 12
		Gold	> 12

low cohesive strength. In polymer-to-mineral adhesion, the weak boundary layer is usually water or a loose corrosion product, resulting from the environment, on the mineral surface. In either case, the adhesive (polymer) must be capable of competing effectively with the potential-weak boundary layers for the substrate surface.

If the surface energy components γ_D and γ_D are known for a substrate and an adhesive, then the thermodynamic stability of the interface between them in the presence of water (an environmental agent of major concern) can be determined. If water diffuses into a joint between two materials A and B, then the work of adhesion can be written (Reference 20) as

$$-\left(y_{p}^{B} y_{p}^{H_{2}0}\right)^{1/2} + \left(y_{D}^{A} y_{D}^{B}\right)^{1/2} + \left(y_{p}^{A} y_{p}^{B}\right)^{1/2}$$
 (3)

The parameter Y^2 is the surface free energy of water and is equal to the sum of Y_D^2 and Y_P^2 , which are 22.0 mJ/m² and 50.2 mJ/m², respectively. If $AB, H_2^0 = W_A^{AB}$, as calculated from Equation 2, then the bond has been weakened by the water. If $W_A = S_A^{AB} = S_A^{$

Using Equation 3, interfacial stability in a moist environment can be predicted graphically for a given substrate as a function of γ_p and γ_p of the adhesive. For example, Figure 7 indicates the regions of stability and instability for bonding to an iron oxide substrate in a wet environment (Reference 20). The line represents those values of γ_p and γ_p for the adhesive that result in a bond strength of exactly zero. It must be kept in mind that this approach to bond durability prediction is thermodynamic in nature and does not include any kinetic considerations. If the time required for water to penetrate the interface is long compared to the life of the module, then even a thermodynamically unstable bond may perform excellently.

To stabilize a joint, then, either water must be prevented from reaching the interface during the module's useful life or the resistance of the interface to water must be increased. One approach to this problem is to seal the joints with materials that have low water-diffusion rates, such as epoxies or phenolic resins. Unfortunately, many of these materials fail to meet one or more requirements for use in solar modules, such as transparency, processibility, etc. The other alternative of stabilizing the interface against attack by water can be achieved by controlled chemical bonding between the

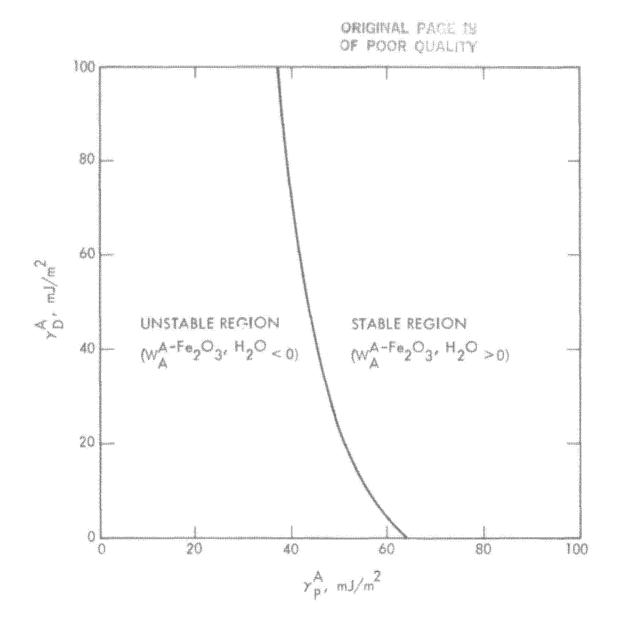


Figure 7. Prediction of Interface Stability for Fe₂O₃ Substrate in a Wet Environment as a Function of $\gamma_{\rm p}$ and $\gamma_{\rm D}$ of Potential Adhesives; Calculation Was Made Using Equation 3 and the Values of 107 mJ/m² and 1250 mJ/m² for $\gamma_{\rm D}$ and $\gamma_{\rm p}$ of Fe₂O₃

adhesive and the substrate through the use of a coupling agent. An excellent example of this approach is seen in fiberglass-reinforced boats, wherein the fiberglass is chemically coupled with silane to the laminating resin. The successful experience of many thousands of such boats in water inspires confidence.

A number of chemically bonding coupling agents have been identified, including modified silanes, titanates, phosphates and chromium complexes. The FSA program has emphasized the use of silane primers because of their water resistance, transparency and ability to bond to a wide variety of materials (organic polymers, metals, metal oxides, glasses, etc.). Furthermore,

organofunctional silane compounds have been used for many years as coupling agents on glass fibers to improve the performance of glass-reinforced polymer composites, and to allow retention of properties upon exposure to moisture. Commercial silane coupling agents are available (Table 11) for bonding to virtually any organic polymer.

The coupling mechanism of adhesion through silanes is complex and only partially understood. The general idea has been to use an ambifunctional silane that will undergo two reactions, one with a mineral substrate and one with an organic adhesive, thus forming a continuous, covalent chemically bonded bridge between the two.

Although the concept that the observed durability of silane-primed interfaces was due to chemical bonding (rather than improved physical adhesion) has existed for many years, only recently has direct evidence of it been available. Ishida and Konig (Reference 21) have used Raman and FTIR spectroscopic techniques to observe coupling-agent-to-substrate interfacial bonds in various glass-silane systems. Largely as a result of their work, the details of bonding to the inorganic substrates are far better understood than those of the bonding to the organic adhesives. The chemistry of a typical silane coupling agent on a silicate glass is outlined in Figure 8 (Reference 22).

Commercial coupling agents are usually available as functionalized trialkoxy compounds. The functional group X is chosen for maximum bonding ability to the organic adhesive. The alkyl groups R (typically methyls) are removed by hydrolysis in hot water and the resulting silanols dissolved in a suitable solvent. The primer is then applied to the clean glass surface and the solvent evaporated, leaving a hydrogen-bonded network. The final step is to cure the system by heating and drying resulting in the formation of a polysiloxane material bound to the glass by means of Si-O-Si bonds. The cured coupling agent shown in Figure 8 is a linear polysiloxane; however, by changing the functionality or mixing multiple silanols, a three-dimensional polysiloxana network can be formed. A major contribution of Konig and coworkers to the study of silane-glass bonding has been the development of a computer-aided FTIR technique with which they can distinguish silicon-oxygen bonds across the interface from those in the polymer. Using this technique, they are able to quantify interfacial bond strengths in systems where mechanical (peel) testing is difficult or impossible, and to identify optimum bonding conditions (Reference 9).

D. INTERPHASE STRUCTURE

It was pointed out in the preceding section that the nature of the coupling agent bonding to an inorganic substrate is fairly well understood. What is still uncertain (and quite controversial) is the nature of the region between the coupling-agent—substrate interface and the bulk organic adhesive, called the interphase. There are at least two models for this region that have attained some acceptance in the adhesion community. However, before examining the details of these different pictures, it is necessary to understand the importance of the interface—interphase region to the integrity of the bond.

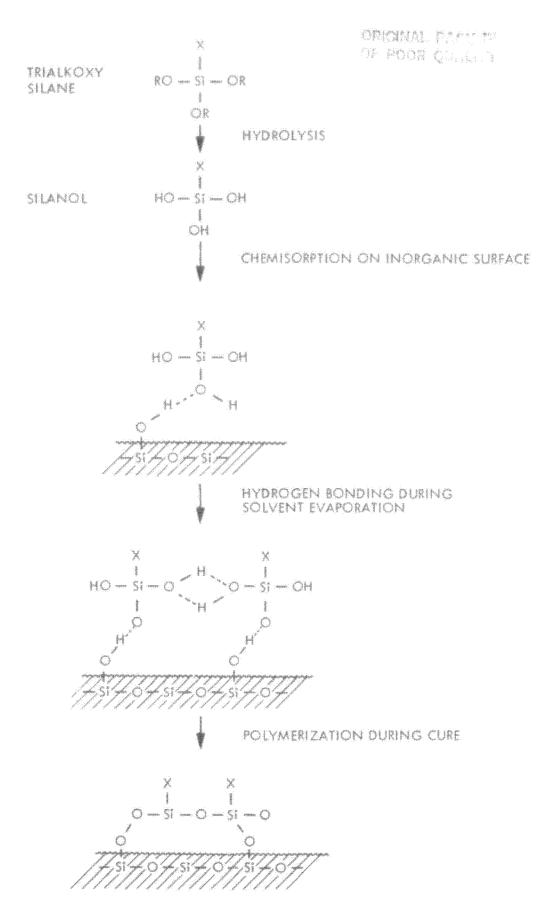


Figure 8. Formation of Polysiloxane on Glass Surface

Table 11. Typical Commercial Coupling Agents and Recommended Applications

Organo functional Dow Corning Silane Designation		Recommended Applications	
Vinylbensyl (cationic)	2 + 6032.	All resins and rubbers	
Vinyl	2-6075	Unsaturated polymers	
Methacrylate (nonionic)	2-6030	Unsaturated polymers	
Methacrylate (cationic)	2-6031	Unsaturated polymers (ketone peroxide)	
Amine	2-6020	Epoxy, phenolic, Nylon	
Ероху	2=6040	Thermosets	
Mercaptan	26062	Thermosets, rubber	

Adhesive joints usually debond as a result of cohesive failure in one of the adherends or in the interphase region itself. Presumably, mechanical strength requirements of materials used in solar modules have been assessed and considered in selection of the structural components (Reference 23). However, even with optimum chemical modification of an interface, a durable, water-resistant bond will not be formed unless the interphase region also has adequate cohesive strength and mechanical properties such as rigidity, tensile strength and toughness to carry the mechanical load when the bonded system is stre.sed. Furthermore, these properties of the interphase must be relatively insensitive to hydrolysis.

On the basis of many observations, a master curve has been developed (Figure 9) relating total performance (including moisture resistance) of composites under stress to the degree of structure in the interphase region of organic-inorganic composites (Reference 7). In all cases, the performance curve of Figure 9 assumes optimum chemical bonding at the interface.

Many of the polymeric materials used in fabricating solar modules have mechanical properties in the range of minimum performance in Figure 9. To develop water-resistant bonds to such polymers, the interfacial region must be modified in such a way as to develop either a tacky (viscoelastic) interphase or a crosslinked resinous interphase. In all cases it is necessary to modify only a very thin region for optimum bonding. This makes it possible to obtain water-resistant bonds of flexible or rubbery coatings to metals and glasses by means of crosslinking without sacrificing total flexibility.

The traditional view of chemically bonded interfaces, coming principally as an extension of physical adhesion ideas, is the sandwich structure shown in Figure 10. This model describes the primed interface (glass-EVA, for example)

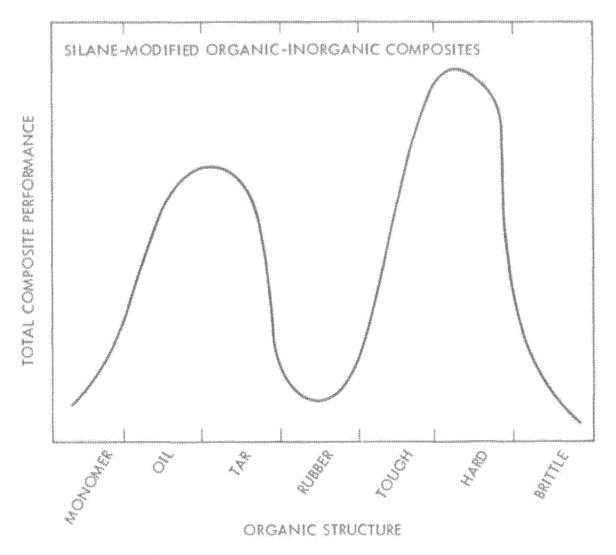


Figure 9. Variation in Total Performance of Organic-Inorganic Composites
With Change in Polymer Morphology Near the Interface

as a series of discrete layers of clearly identifiable composition and structure. These layers include:

- (1) The bulk glass substrate.
- (2) A region including the surface monolayer of glass molecules and the contacting monolayer of polysiloxane molecules between which there are chemical bonds (interface 1).
- (3) A region of bulk polysiloxane (interphase).
- (4) A region including the surface monolayer of EVA molecules and the contacting monolayer of functionalized polysiloxane molecules between which there are chemical bonds (interface 2).
- (5) The bulk EVA region.

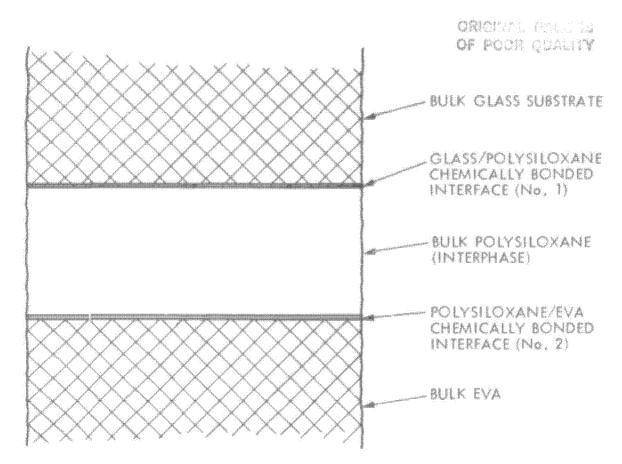


Figure 10. Sandwich Model of Silane-Primed Glass-EVA Joint

Within the framework of the sandwich model, two clearly defined interfaces and an interphase region consisting of bulk polymerized coupling agent can be identified. If this is an accurate picture of the bond, then, assuming adequate chemical bonding strength at the interface, the performance of bonds in solar modules using silane primers will depend on the performance of the bulk polysiloxane interphase.

The properties of the polysiloxane layer are primarily dependent on the nature and extent of the curing (polymerization) process. The initially deposited silanols must be crosslinked in order to form a water-resistant bond (Reference 24). Before or during the early stages of cure, the bulk primer region consists of a multilayered structure of oligomeric siloxanels that are soluble and fusible. As heating continues, this condenses into a crosslinked silicone structure that is insoluble and infusible.

The ability of such a network to absorb and dissipate energy when stressed is a function of a number of material properties. Two of these, the glass transition temperature (T_g) and the fracture energy, are of primary importance.

It has been suggested that a good criterion for the T_g of the interphase is that it be in the middle of the service temperature range of the modules. This is based on the fact that a material's ability to dissipate low-frequency

stresses, which are presumably the most important in solar modules, is maximum near T_g . The T_g of the polysiloxane will be a function of the processing conditions. Kaelble (Reference 25) and coworkers at Rockwell International have carried out some preliminary studies on the T_g of several cured silane primers (DC-26020, DC-26030 and DC-26031) by thermal mechanical analysis. Silanes were mixed with acidified water (pH 4) in a weight ratio of 50:50 and allowed to polymerize and evaporate, simultaneously, by heating at 130°C overnight. Small sample pieces were then removed and T_g was measured (at a scan rate of 10°C/min). After scanning to about 400°C , samples were cooled and rescanned to determine the effect of the first scan (postcuring) on the T_g of the silanes.

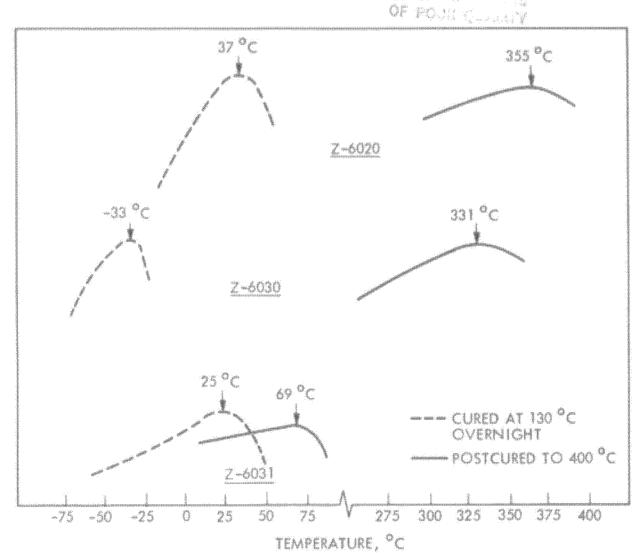
Figure 11 shows the glass transition temperatures of the silanes after polymerization at 130°C and after thermal scanning to about 400°C . The data show that silanes polymerized at 130°C (as recommended in customary processing conditions) have rather low T_g (i.e., 37°C , -33°C , and 25°C for Z-6020, Z-6030, and Z-6031, respectively). However, postcuring raised the T_g substantially, especially for silanes Z-6020 and Z-6030. This increase in T_g is conceivably due to additional network formation or network rearrangement as a result of postcuring.

The criterion of high fracture energy is more difficult to meet in the case of the silane coupling agents than that of $T_{\rm g}$; bulk polymerized silanes are known generally to possess little mechanical strength. They are, in fact, widely used as mold release agents for high-temperature curing processes.

Critics of the sandwich model argue that if there is a layer of bulk polysiloxane present at an interface, then the bond will be inherently weak because of the low cohesive strength of this material. Recently, experimental evidence that appears to support this argument has come to light. Studies by Koenig (Reference 9) of bond strengths of epoxy-glass is verfaces costed with silane primers show that as the amount of primer on the surface of the glass before bonding is increased, the bond strength initially rises, reaches a maximum and then declines. The interpretation of these results is that the initial increase in bond strength is due to increasing density of interfacial chemical bonds. Once a chemically bound monolayer and a few (probably <10) strongly chemiabsorbed layers of coupling agent have been deposited on the glass surface, addition of further primer only results in the formation of a bulk silanol layer which, due to its poor mechanical properties after cure, is subject to cohesive failure, resulting in a lowered bond strength (Figure 12) (Reference 26). If excess primer is deposited on the surface, it can be removed by washing with hot water or alcohol before bonding, with a resulting increase in bond strength being realized. It is apparent that this phenomenon is not limited to silane primers. Matsucka (Reference 27) has reported a similar bond strength dependence on adhesive thickness in studies of Al/Nylon 12 laminates.

Recently, an alternative picture of the chemically bonded interface and the interphase region in particular has begun to emerge. The interdiffusion model, as we shall call it, is depicted in Figure 13 for a silane-primed glass-EVA interface. This model differs from the sandwich model in that there is no bulk coupling agent layer present, nor is there a clearly defined coupling-agent-organic interface. Rather there is an interphase region where



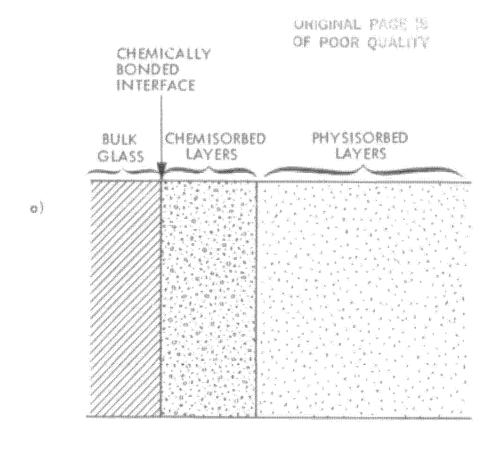


Ordor

Figure 11. Glass Transition Temperatures of Silane Coupling Agents

interdiffusion has occurred between the coupling agent and the organic adhesive before cure. The composition of this region as a function of distance from the silane-glass interface is shown schematically in Figure 14.

To establish a density gradient of the organic adhesive in the interphase region, it is necessary that it be soluble in the coupling agent or primer solution and that at some time during the bonding process these two materials have low enough viscosity to allow mixing. Once the materials have mixed, the system can be cured, resulting in the formation of a heterogeneous network. Proponents of the interdiffusion model argue that such a network will be much stronger than a pure bulk polysiloxane layer and that it is this type of structure that is responsible for the high bond strengths observed in chemically bonded organic-inorganic interfaces.



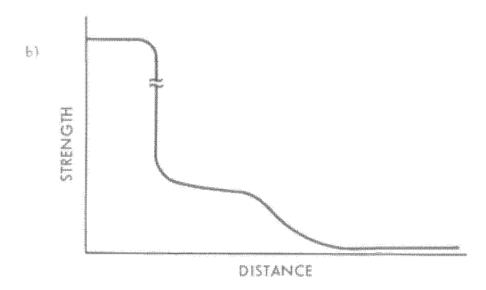


Figure 12. Silane Coupling Agent on Glass: (a) Schematic Representation Showing Chemically Bonded Interface, Strongly Chemisorbed Region and Weakly Physisorbed Region; (b) Plot of the Mechanical Strength of Various Regions Near the Interface, Showing Weakness in the Physisorbed Bulk Silane

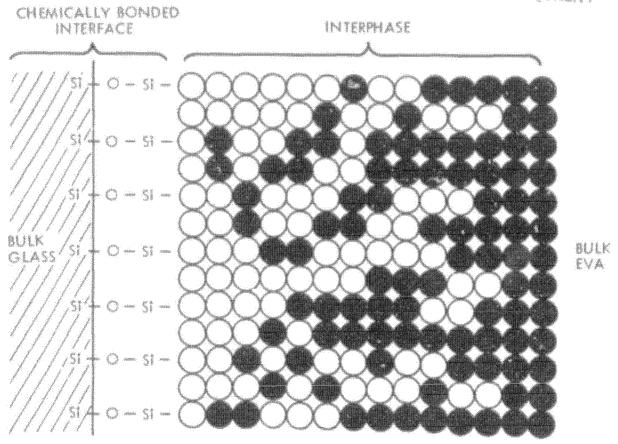


Figure 13. Schematic Representation of the Interdiffusion Model for a Silane-Primed Glass-EVA Joint; Open Circles Indicate Regions of Coupling Agent, Filled Circles Indicate Regions of EVA

The structural details of the interphase region are largely unknown. However, depending on the nature of the coupling agent and the organic adhesive, one of four basic structures is likely. In bonding to reactive thermoset resins, the coupling agent may diffuse into the resin and copolymerize during cure, forming a very strong chemically bonded network. In less reactive resins, partial diffusion may occur followed by separate polymerization of the coupling agent and the organic adhesive, resulting in the formation of an interpenetrating network. In the case of non-reactive thermoplastics such as polyethylene or polypropylene, after interdiffusion only the coupling agent will crosslink, leaving a pseudo-interpenetrating polymer network. Finally, in some cases long-chain linear coupling agents will be used, which will interdiffuse with the polymer matrix but will not crosslink, leaving either a pseudo-interpenetrating network or simply a heterogeneous mixture of long-chain polymers, depending on the nature of the organic adhesive.

Studies of bonding of silane-primed glass/EVA joints appear to support the interdiffusion model. If the silane coupling agents used cure before melting of the EVA, there is little chance for interdiffusion and the bonding

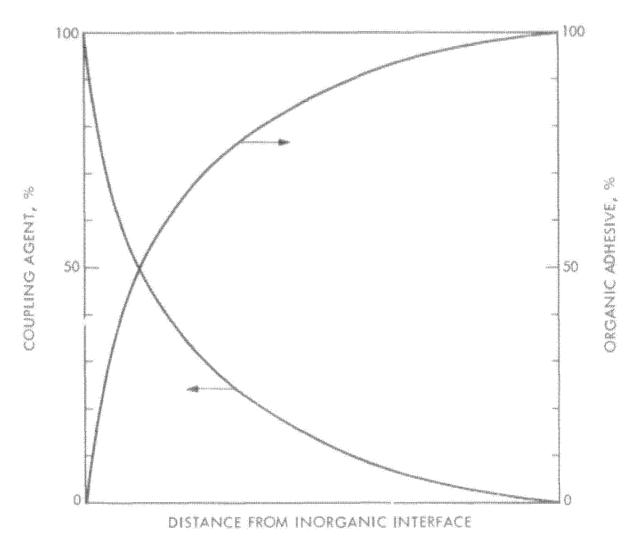


Figure 14. Plot of Silane-Adhesive Concentration Gradient as a Function of Distance From Inorganic Interface

is weak. It is observed, nowever, that bond strengths improve if silanols that cure at a temperature similar to that of EVA are used, allowing interdiffusion of the two polymers. Furthermore, if a crosslinking agent (Lupersol 101) is added to the EVA, an interpenetrating network can be formed with further significant increases in bond strength being realized.

At the time of this report, the theories of interfacial bonding are in a state of flux. Many of the new concepts being proposed and discussed are not yet proven experimentally. However, considerable work is being done in these areas and, it is hoped, in the near future the picture will become much clearer.

SECTION IV

PROCESSING CONSIDERATIONS RELEVANT TO INTERPACTAL BONDING

Even if optimum systems for bonding in solar modules are identified in the laboratory, strong durable interfacial bonds may not be obtained if careful consideration is not given to processing techniques. Development of production guidelines to insure adequate bond performance is not complete; hever, a growing body of experience and research has identified some basic principles.

Processing considerations of importance to interfacial bonding fall into three broad categories:

- (1) Surface preparation.
- (2) Application.
- (3) Curing.

In this section, some (but certainly not all) of the processing issues of relevance to bonding will be discussed. In many instances the issues will be presented for information, without specific recommendations, due to a lack of sufficient data.

A. SURFACE PREPARATION

Preparation of adherend surfaces before application of a primer or adhesive is of fundamental importance. Unless weakly bound surface layers are removed, they will likely lead to failure of the joint. The surface layers may be water, oil, organics, oligomeric species, corrosion products or other contaminants. For a given surface, the nature of these layers must be determined and suitable cleaning procedures (abrasion, washing, etching, etc.) established. Furthermore, the time between cleaning and application of the bonding materials should be kept to a minimum to avoid recontamination.

Most inorganic materials also have oxide and/or hydroxide layers on their surfaces. In a humid environment, the durability of clean glass or silicon oxide surfaces of solar cells is controlled by the reaction

$$Si-O-Si + H_2O \longrightarrow 2 SiON.$$
 (4)

Landsford (Reference 28) has given an expression for the depth (D) of hydroxyl converge of glass as follows:

$$p^2 = x \epsilon \tag{5}$$

where t is time in years and K = $3.3 \times 10^{-9} \, \mathrm{m}^2/\mathrm{yr}$ is the diffusion rate constant for water in glass. Using Equation 5, a simple calculation shows that a 1-Å-thick hydroxide layer will build on a glass surface in approximately 11 days. In order to eliminate completely the surface hydroxide

layer on E-glass fibers used in glass-reinforced epoxy, they are coated with primer immediately (<1 sec) after being drawn.

It is important to understand the effects of the presence of oxidized or hydrolyzed surface layers on bonding to determine whether it is necessary or desirable to minimize them. Generally, it is found that bonding to a glass surface is more efficient if the hydroxide layer is removed. The presence of such a layer may adversely affect the bond strength by causing the coupling agent to orient itself in a less than optimum geometry on the surface (Reference 29). Removal of the hydroxide layer can be achieved by etching the glass in a strong acid solution. On the other hand, there is some suggestion (Reference 30) that treatment of metal surfaces should maximize surface hydroxylation to produce optimum bonding. Subsequent chemical reaction with low-pH groups of adhesives and coupling agents seems to provide durable, corrosion-resistant interfaces.

B. APPLICATION

Once the adherend surfaces have been properly cleaned, they are ready for application of the coupling agents and/or adhesives. These are usually applied as neat liquids or solutions. The physical methods of application may include dipping, spraying, wiping, painting, etc. The method of choice will be based on the nature (viscosity, temperature, reactivity, etc.) of the liquids being applied.

If a material is not applied as a neat liquid then care must be taken in choosing a solvent and preparing the solution. If we consider the preparation of a primer, the first and most obvious criteria for selection of a solvent are that it must dissolve the coupling agents and that it must be non-reactive with them. It must also be capable of wetting the surfaces of the adherends, as discussed in Section III.B above, and it should be sufficiently volatile to evaporate in a reasonable amount of time under relatively mild conditions. The pH of the solution may also need to be adjusted, since it may affect the stability of the coupling agents as well as their deposition on the adherend surfaces.

Boerio and coworkers (Reference 31) have defined specific pH effects on aminosilane orientation on oxidized iron with an isoelectric point of pH \cong 8.5. When the aminosilane is applied from basic solution, pH \cong 10.5, there is a strong tendency for the molecules to deposit upside down (with the amino groups next to the oxide). At lower pH there is a reported tendency to reverse this orientation with silanols bonding to the oxide. At high pH the tendency to form the ring structures of amine to silane is enhanced and lowering the pH tends to open the rings and enhance the amine reactivity with the organic resin phase.

In Section III it was emphasized that the thickness of the coupling agent layer is critically important to bonding. In practice, the amount of coupling agent deposited on the surface is most easily controlled by adjusting the concentration of the primer solution. However, the concentration must never be allowed to reach such a level that molecular association begins to occur. It has been found that under such conditions the strength and durability of

the interface degrades (Reference 32). After deposition and initial drying of the primer, it may be desirable to remove some excess coupling agent before application of the adhesive and cure. This can be achieved by washing the surface in an appropriate solvent (e.g., hot water, alcohol) followed by a second drying.

C. CURE

The final category of processing considerations relevant to bonding involves curing of the coupling agents and adhesives. The physical and mechanical properties of the bulk adhesive and the interphase region are of critical importance to maintaining a strong bond. Most of these properties, such as Tg, fracture energy, crosslink density, network topology, etc., are strongly dependent on the details of cure, and slight changes in the curing cycle can alter the materials significantly.

Cure chemistry of multifunctional, multicomponent systems is so complex that it is almost impossible to predict the relationship of cure cycle to final mechanical properties; consequently, studies of these relationships are virtually all empirical in nature. For this reason, it is almost impossible to give even rudimentary guidelines for cure of bonded interfaces. The guiding principle, however, must be that the cure produces an interphase region with mechanical properties that give maximum performance in terms of bond stability and durability (Section III.D.).

- 1. Willis, P., et al., Investigations of Test Methods, Material Properties and Processes for Solar Cell Encapsulants, Annual Report, ERDA/JPL-954527, Springporn Laboratories, Inc., Enfield, Conn., July, 1977.
- 2. Willis, P., Baum, B., and White, R., Investigations of Test Methods, Material Properties and Processes for Solar Cell Encapsulants, Annual Report, ERDA/JPL-954527, Springborn Laboratories, Inc., Enfield, Conn., July, 1978.
- 3. Gupta, A., Photodegradation of Polymeric Encapsulants of Solar Cell Modules, JPL Internal Document No. 5101-77, Jet Propulsion Laboratory, Pasadens, California, August 10, 1978.
- Gupta, A., Effect of Photodegradation on Chemical Structure and Surface Characteristics of Silicone Pottants Used in Solar Cell Modules, JPL Internal Document No. 5101-79, Jet Propulsion Laboratory, Pasadena, California, August 18, 1978.
- 5. Runge, M.L., and Dreyfuss, P., J. Polymer Sci., 17, 1067, 1979.
- b. Pineddemann, E.P., Silane Coupling Agents, Plenum Press, New York, 1982.
- 7. Plueddemann, E.P., Chemical Bonding Technology for Terrestrial Solar Cell Modules, JPL Internal Document No. 5101-132, Jet Propulsion Laboratory, Pasadena, California, September 1, 1979.
- 8. Willis, P., and Baum, B., <u>Investigation of Test Methods</u>, <u>Material Properties</u>, and <u>Processes for Solar Cell Encapsulants</u>, <u>Annual Report</u>, <u>DOE/JPL-954527-82/83</u>, <u>Springborn Laboratories</u>, <u>Inc.</u>, <u>Enfield</u>, <u>Conn.</u>, July, 1982.
- 9. Koenig, J., FSA Consulting Contract KO-740528, with the Jet Propulsion Laboratory, Pasadena, California private communication.
- 10. Willis, P., and Baum, B., Investigations of Test Methods, Material Properties and Processes for Solar-Cell Encapsulants, Annual Report, Springborn Laboratories, Inc., Enfield, Connecticut, DOE/JPL 954527-80-15, Jet Propulsion Laboratory, Pasadens, California, July 1981.
- 11. Willis, P., et. al., Investigations of Test Methods, Material Properties and Processes for Solar-Cell Encapsulants, Annual Report, ERDA/JPL-954527, Springborn Laboratories, Inc., Enfield, Connecticut, July, 1980.
- 12. White, M.L., "Encapsulation of Integrated Circuits," Proceedings of the IEEE, Vol. 57, p. 1610, 1969.
- 13. Jaffe, D., "Encapsulating Integrated Circuits Containing Beam Leaded Devices with a Silicone RTV Dispersion," IEEE Transaction on Parts, Hybrids, and Packaging, Vol. 12, p. 182, 1976.



- 14. White, M.L., "Encapsulating Integrated Circuits," Bell Laboratories Record, Vol. 52, p. 80, March 1974.
- 15. Shar, N.L., and Feinstein, L.G., "Performance of New Copper-Based Metallization Systems in an 85°C, 80% Rh, Gl₂ Contaminated Environment," Proceedings of 1977 Electronic Components Conference, Arlington, Virginia, May 16-18, 1972, pp. 84-95.
- 16. Kinloch, A.J., J. Mat. Sci., 15, 2141, 1980.
- 17. Kaelbe, D.H., Mansfield, F.B., and Kendig, M., Study Program for Encapsulation Materials Interface for Flat-Plate Solar Arrays, Annual Report, JPL-954739, Rockwell Science Center, Thomsand Oaks, California, February, 1982.
- 18. Parks, G.A., Chem. Rev., 65, 177, 1965.
- 19. Bolger, J.C., and Michaels, A.S., Interface Conversion, pp. 3-51, ed. P. Weiss, Elsevier, Amsterdam, 1968.
- 20. Kinloch, A.J., J. Mat. Sci., 17, 617, 1982.
- 21. Ishida, H., and Koenig, J., "New Spectroscopic Techniques for Studying Glass Surfaces," <u>Sylated Surfaces</u>, ed. D.E. Leyden and W.J. Collins, Gordon and Breach, New York, 1980.
- 22. Lee, L.H., J. Coll. and Interface Sci., 27, 751, 1968.
- 23. Guddihy, E.F., Carroll, W., Coulbert, C.D., Gupta, A., and Liang, R., Photovoltaic Module Encapsulation Design and Materials Selection: Vol. I, JPL Publication 81-102, Jet Propulsion Laboratory, Pasadens, California, June 1, 1982.
- 24. Plueddemann, E.P., "Chemistry of Silane Coupling Agents," Sylated Surfaces, ed. D.E. Leyden and W.T. Collins, Gordon and Breach, New York, 1980
- 25. Kaelble, D.H., FSA Subcontract No. 954739, Monthly Progress Report No. 58, November 16, 1982.
- 26. Ishida, H., "Structure of Physisorbed and Chemisorbed Silane Coupling Agents and Their Role in the Reinforcement Mechanisms and Processibility," presented at the 185th American Chemical Society National Meeting, Seattle, Washington, March 20-25, 1983.
- 27. Matsuoka, M., Int J. of Adhesion, 2, 223, 1982.
- 28. Landsford, W.A., Science, 196, 975, 1977.
- 29. Zettlemoyer, A.C., and Hsing, H.H., J. Col. and Interface Sci., 58, 263, 1977

- 30. Kaelble, D.H., "Principles and Practices of Adhesion," presented at the Critical Photovoltaic Technology Workshop of the Flate-Plate Solar Array Project, Pasadens, California, November 10, 1981.
- 31. Boerio, P.J., Armogan, L., and Cheng, C.Y., J. Col. and Interface Sci., 73, 416, 1980.
- 32. Ishida, H., "Molecular Structure of Interfaces in Composite Materials," presented at the 185th American Chemical Society National Meeting, Seattle, Washington, March 20-25, 1983.